



**Lomonosov
Moscow State
University**

**Trilateral workshop on Hot
Topics in HTSC: Fe-Based
Superconductors**

New members of Fe-based superconducting family $A_xFe_{2-y}Se_2$: synthesis, microstructure and properties

**Zvenigorod,
September 29 – October 2, 2013**

Outline

I. Introduction

II. A comparative study of the microstructure of the superconducting and non-superconducting samples



III. Synthesis and study of the properties ferroselenids $(\mathbf{Na,K})_x\mathbf{Fe}_{2-y}\mathbf{Se}_2$ with partial substitution of K by Na.

IV. Synthesis of ferroselenids $\mathbf{A_x(C_5H_5N)_yFe_{2-z}Se_2}$
 $(\mathbf{A=Li, Na})$ by intercalation of lithium and sodium to FeSe in pyridine.

Classification of layered Fe-containing superconductors

Family	Parent compound	Substitution	Examples	T_c max, K
1111	$REOFeAs$ $RE=La,Ce,Pr,Nd,Sm,Tb,Dy$ $AEFFeAs, AE=Ca, Sr$	$O \rightarrow F$	$LaO_{1-x}F_xFeAs$	55
		$AE \rightarrow RE$	$Ca_{1-x}La_xFFeAs$	36
122(As)	$AEFe_2As_2$ $AE=Ca, Sr, Ba, Eu$	$AE \rightarrow A$	$Ba_{1-x}K_xFe_2As_2$	38
		$Fe \rightarrow Co$	$SrFe_{1.7}Co_{0.3}As_2$	30
		$As \rightarrow P$	$BaFe_2(As_{2/3}P_{1/3})_2$	33
		$AE \rightarrow RE$	$Ca_{1-x}La_xFe_2As_2$	47
122(Se)	$A_{0.8}Fe_{1.6}Se_2$ $A=$ alkali and alkaline earth metals		$K_xFe_{2-y}Se_2$	31
			$Li_x(NH_2)_y(NH_3)_{1-y}Fe_2Se_2$	44
			$Li_x(C_5H_5N)_yFe_{2-z}Se_2$	44
111	$LiFeAs,$ $NaFeAs$	$Fe \rightarrow Co$	$LiFeAs,$	18
			$Na_{1-x}(Fe,Co)As$	23
11	FeSe	$Se \rightarrow Te$	$Fe_{1+x}Se$ $FeSe_{0.4}Te_{0.6}$	9 15

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		$AE \rightarrow RE$	$Ca_{1-x}La_xFFeAs$	36
122(As)	$AEFe_2As_2$ $AE=Ca, Sr, Ba, Eu$	$AE \rightarrow A$	$Ba_{1-x}K_xFe_2As_2$	38
		$Fe \rightarrow Co$	$SrFe_{1.7}Co_{0.3}As_2$	30
		$As \rightarrow P$	$BaFe_2(As_{2/3}P_{1/3})_2$	33
		$AE \rightarrow RE$	$Ca_{1-x}La_xFe_2As_2$	47
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			$Li_x(NH_2)_y(NH_3)_{1-y}Fe_2Se_2$	44
			$Li_x(C_5H_5N)_yFe_{2-z}Se_2$	44
111	$LiFeAs,$ $NaFeAs$	$Fe \rightarrow Co$	$LiFeAs,$	18
			$Na_{1-x}(Fe,Co)As$	23
11	FeSe	$Se \rightarrow Te$	$Fe_{1+x}Se$ $FeSe_{0.4}Te_{0.6}$	9 15

Key features of 122(FeSe) family

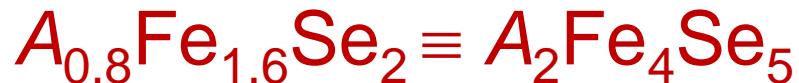
1) Variable composition with a deficit in the cation sublattices of iron and alkali metal:



2) The formal oxidation state of iron in SC-samples is close to **2**. This means that with increasing iron content (2-y) the alkali metal content x decreases, and vice versa.

3) Depending on the composition, synthesis conditions and temperature measurements can be realized different micro-(nano)structures as a result of different ways of ordering the iron and alkali metal.

4) Parent compound for $A_xFe_{2-y}Se_2$ family is

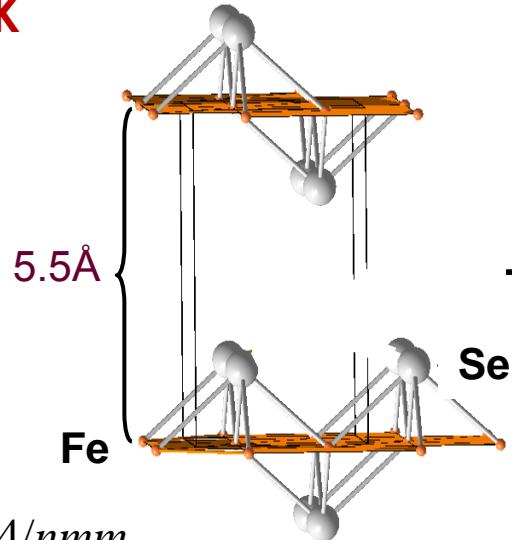


20% of Fe and K atomic positions are empty. As rule substances with small deviation from this composition may be superconductors.

Synthesis of $A_xFe_{2-y}Se_2$ by intercalation of t-FeSe

[J.Guo, S.Jin, G. Wang, et al PRB, 82, 180520R (2010)]

$T_C=9K$



Пр. гр.*P4/nmm*

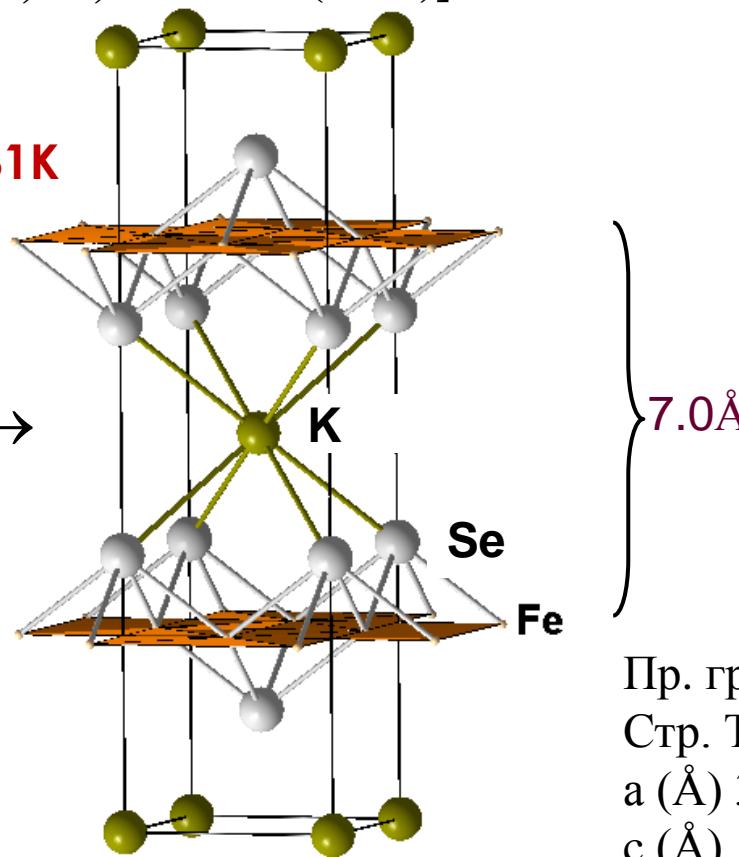
Стр. Тип *анти-PbO*

a (Å) 3.77376(2)

c (Å) 5.52482(5)

$T_C=31K$

+ K →



Пр. гр.*I4/mmm*

Стр. Тип $ThCr_2Si_2$

a (Å) 3.9136(1)

c (Å) 14.0367(7)

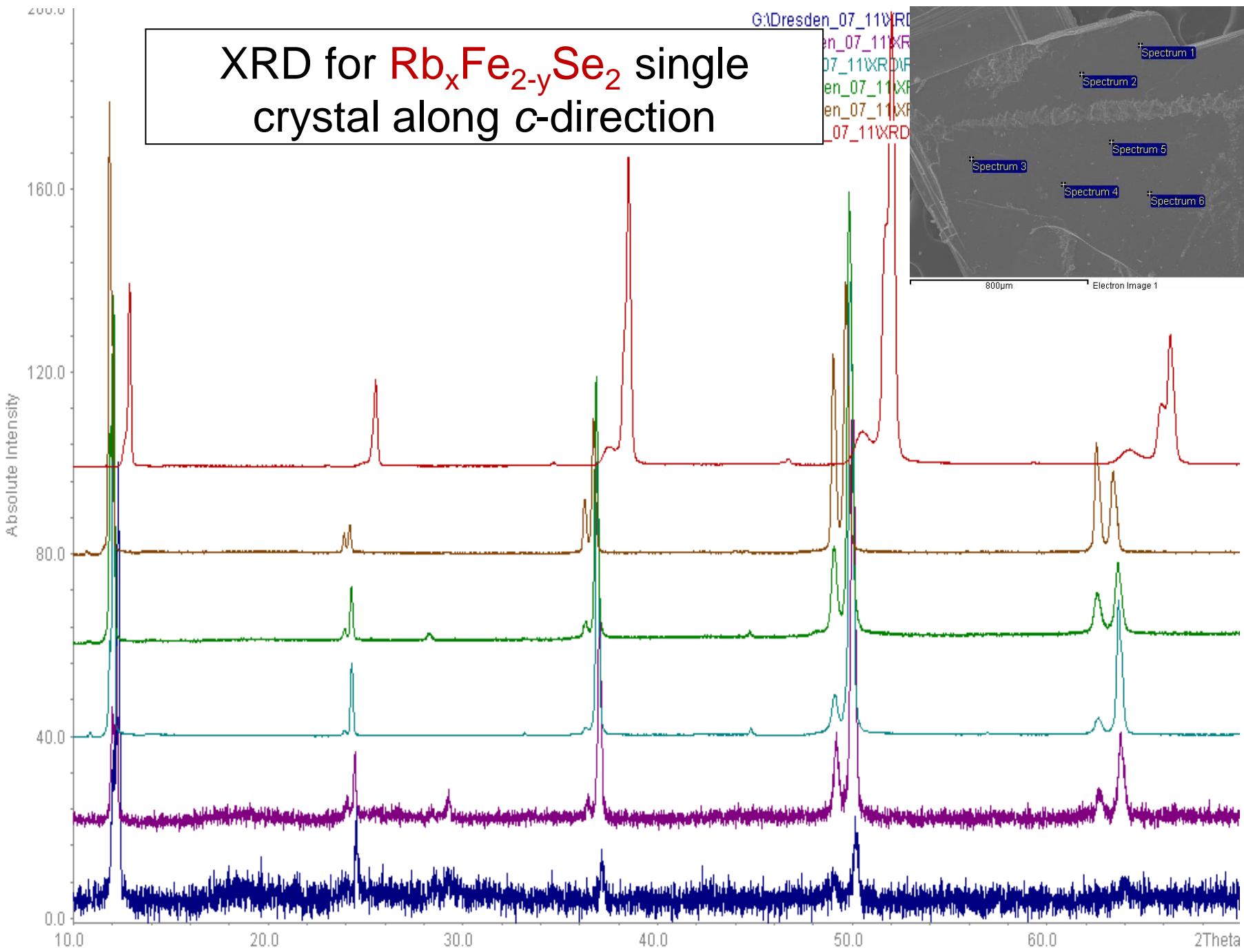
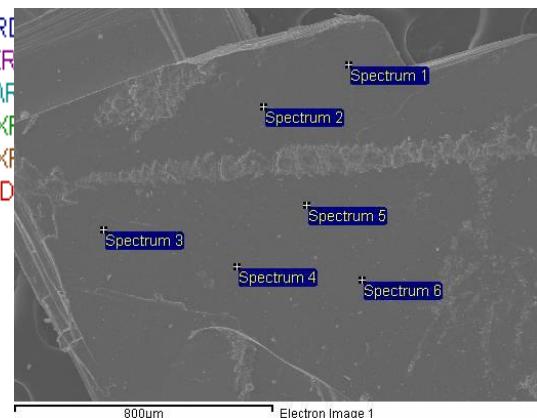
Synthesis by “self-flux” technique:

(2FeSe+0.8K) in double quartz tube, heating up to 1030-1070°C, short dwell (2-5hs) and cooling with the rate about 5-6°C up to 750°C

XRD for $\text{Rb}_x\text{Fe}_{2-y}\text{Se}_2$ single crystal along c-direction

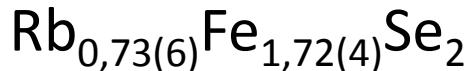
G:\Dresden_07_11\XRD

en_07_11XRD
07_11XRDIF
en_07_11XP
en_07_11XP
07_11XRD

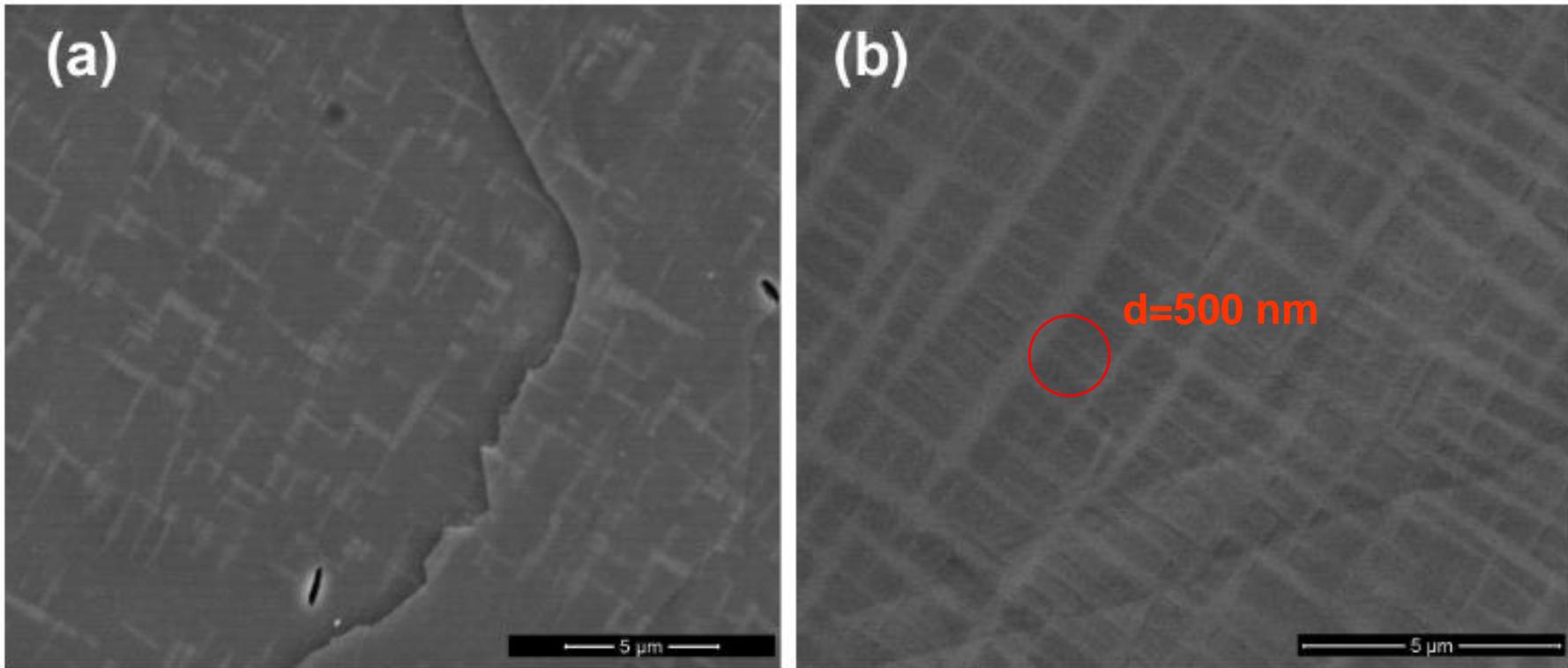


SEM and EDX

The SC sample (**Tc 24K**)



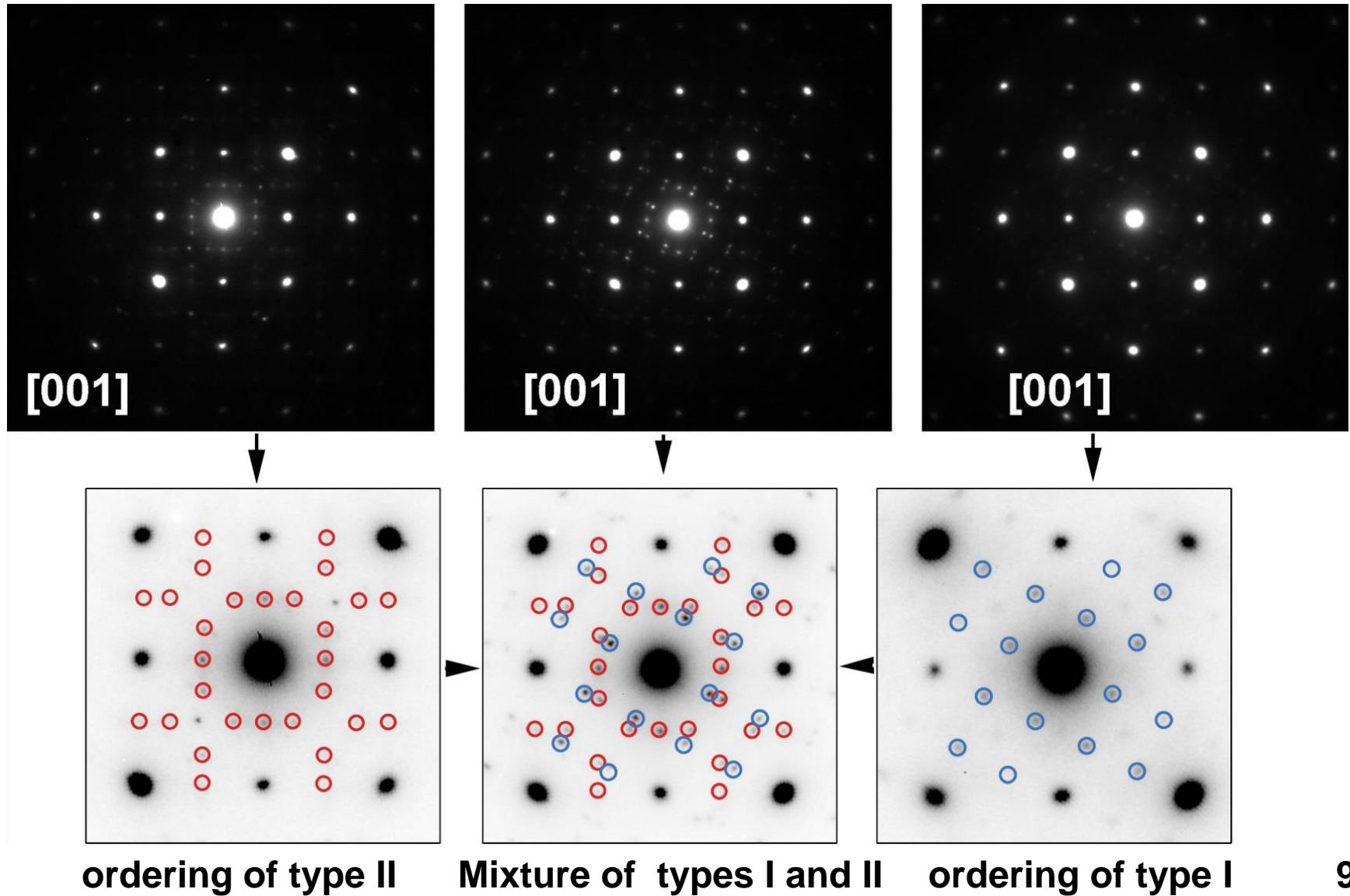
The non-SC sample

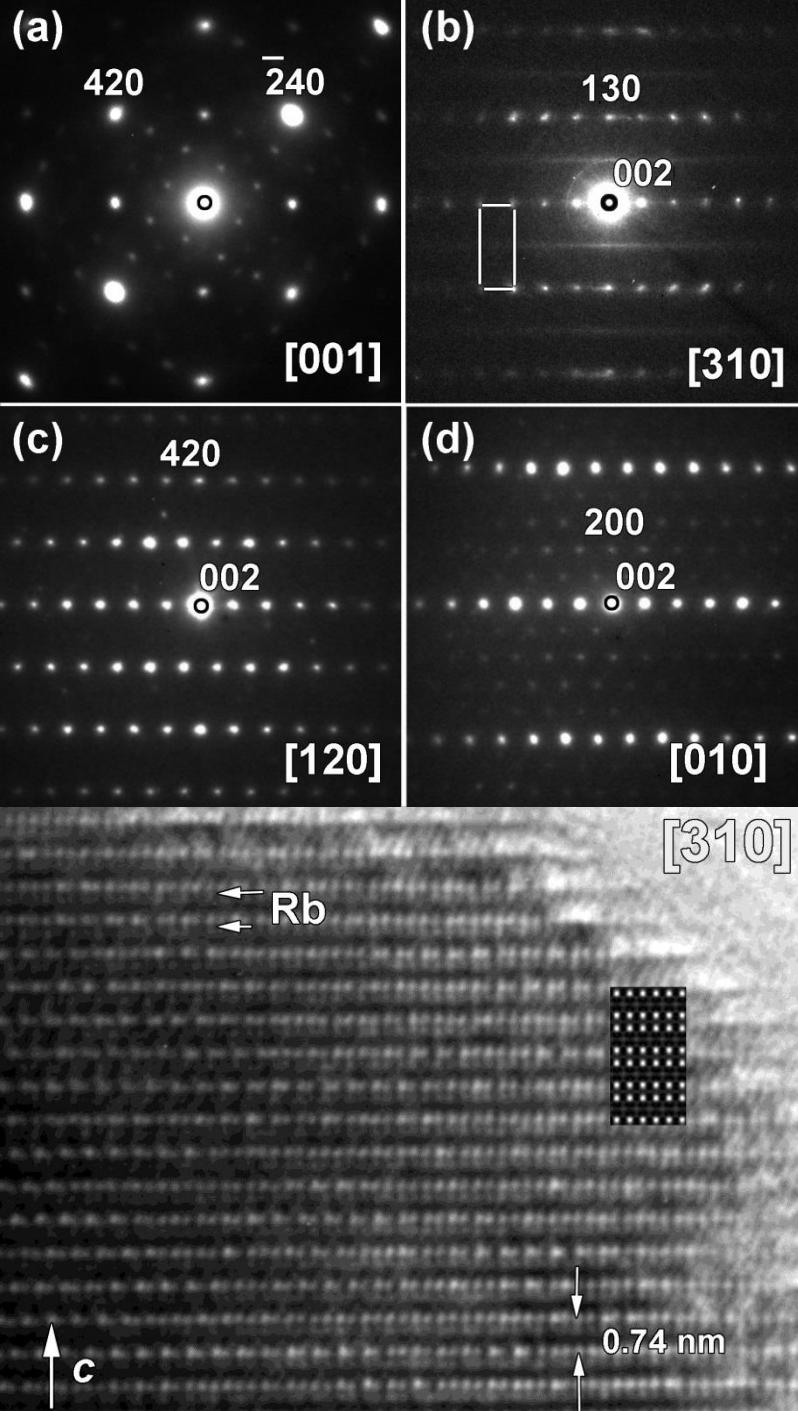


The phase contrast indicates a microinhomogeneity of the samples

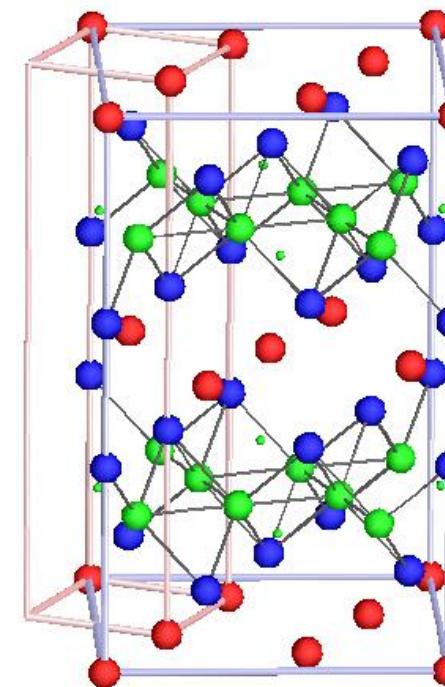
Thus, in this work we solved the problem of a detailed study of the microstructure of the superconducting and non-superconducting sample $\text{Rb}_x\text{Fe}_{2-y}\text{Se}$ to identify the structural features responsible for the superconducting properties.

The variety of superstructures in the [001] zone-axis direction





Superstructure I: the well-known Fe vacancy-ordered structure (space group $I\bar{4}/m$)



$$a_I = 2a_s + b_s$$

$$b_I = -a_s + 2b_s$$

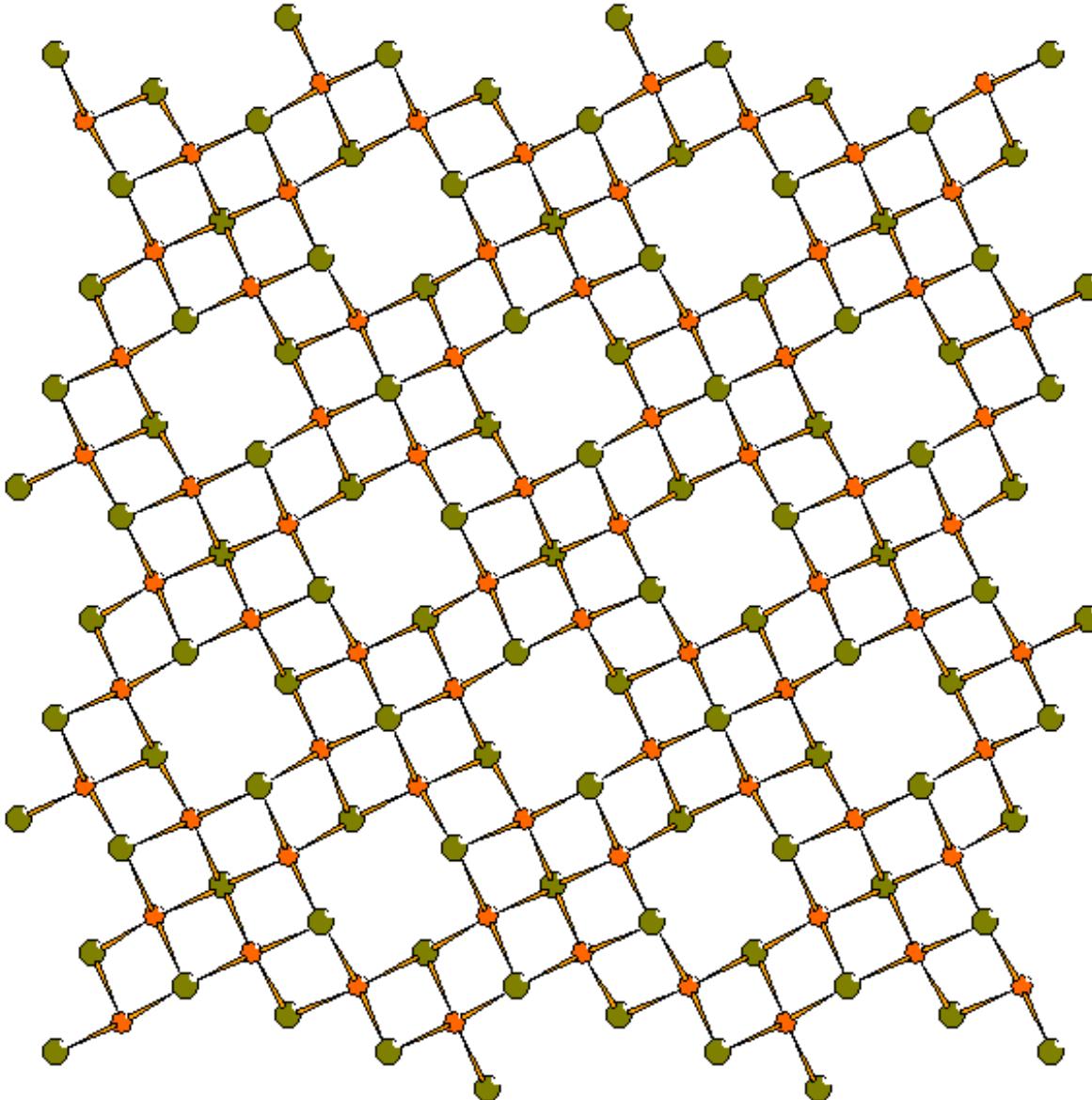
$$c_I = c_s$$

$$a_I = b_I = a_s \sqrt{5} \approx 8.7 \text{ \AA}$$

$$c_I = 14.5 \text{ \AA}$$

$K_{0.8}Fe_{1.6}Se_2$: 1/5 part of Fe atoms absent.

Ordered structure: Space group $I4/m$ ($T < 580$ K)

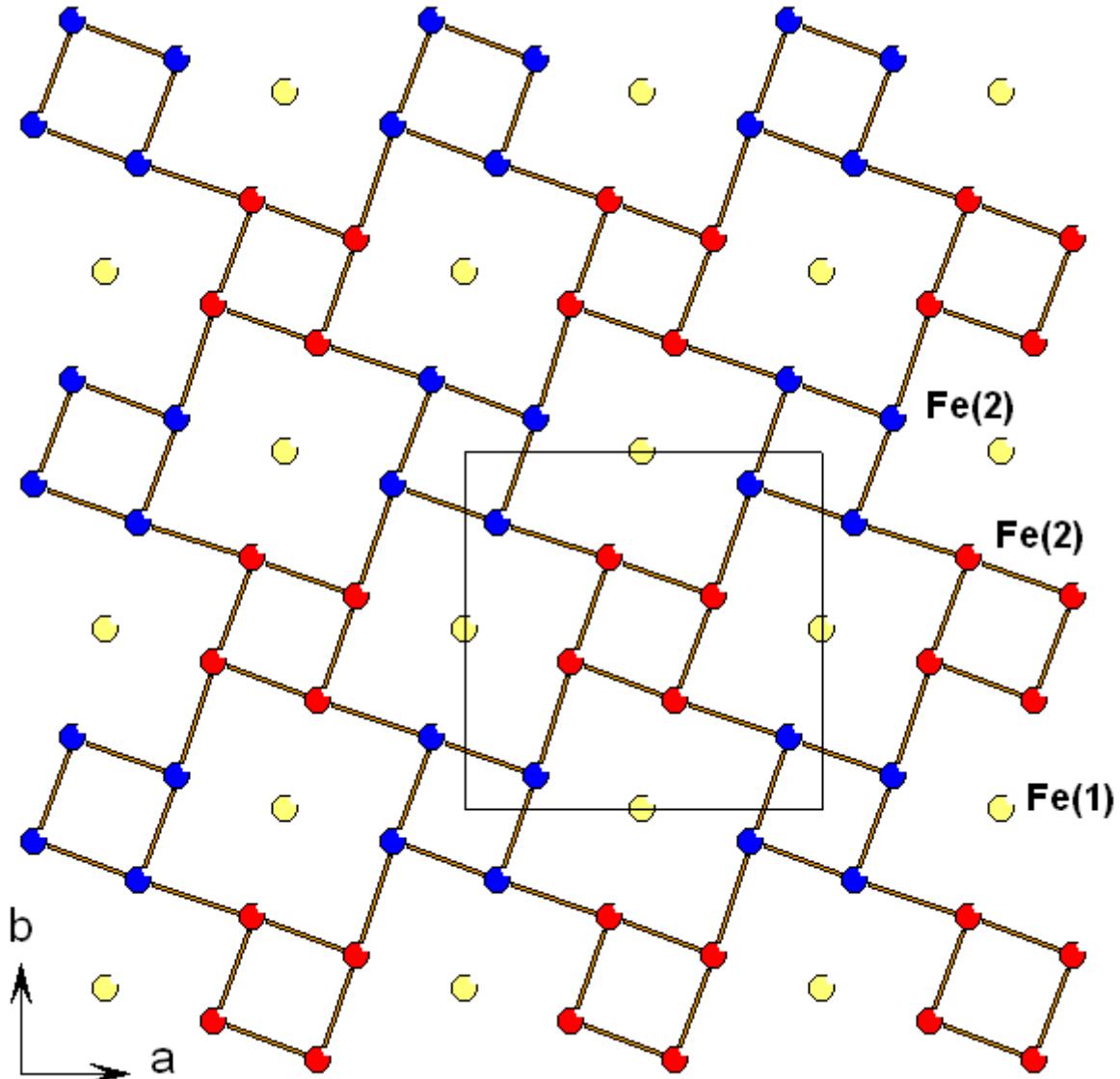


With decreasing temperature, the ordering of vacancies occur.

Legend:

- - Fe
- - As

$K_{0.8+x}Fe_{1.6+y}Se_2$ *I4/m* structure: antiferromagnetic ordering at the $T_N < 560$ K



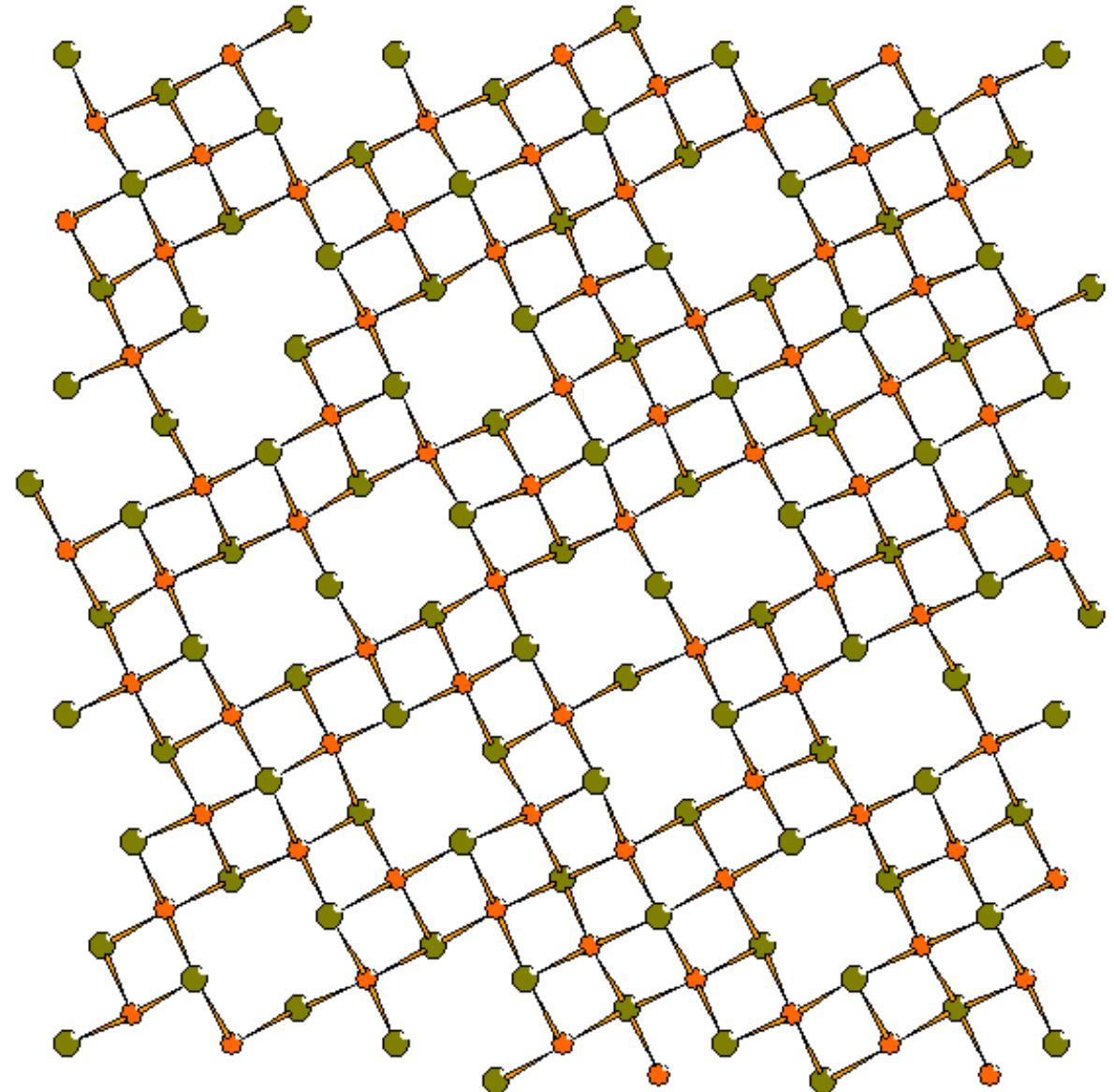
For clarity, only Fe atoms
in one layer (without Se)
are shown

[Bao, W. et al. 2011, arXiv:1102.3674]
(*Neutron powder diffraction*)

Legend:
● and ● - spin-up and
spin-down configurations
of Fe(2) atoms

$K_{0.8}Fe_{1.6}Se_2$: 1/5 part of Fe atoms absent.

Disordered structure: Space group $I4/mmm$ ($T > 580$ K)



At the temperature more than 570 – 600K vacancies of Fe atoms are distributed in layers without ordering.

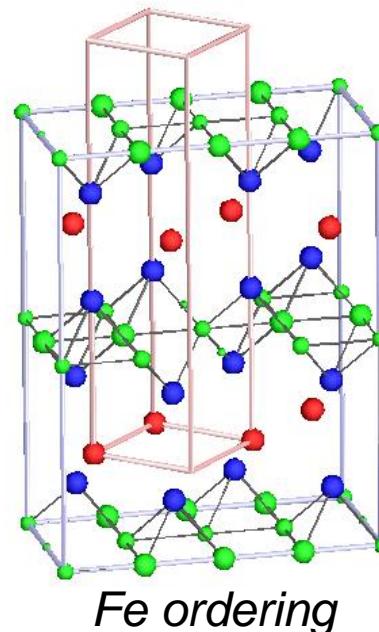
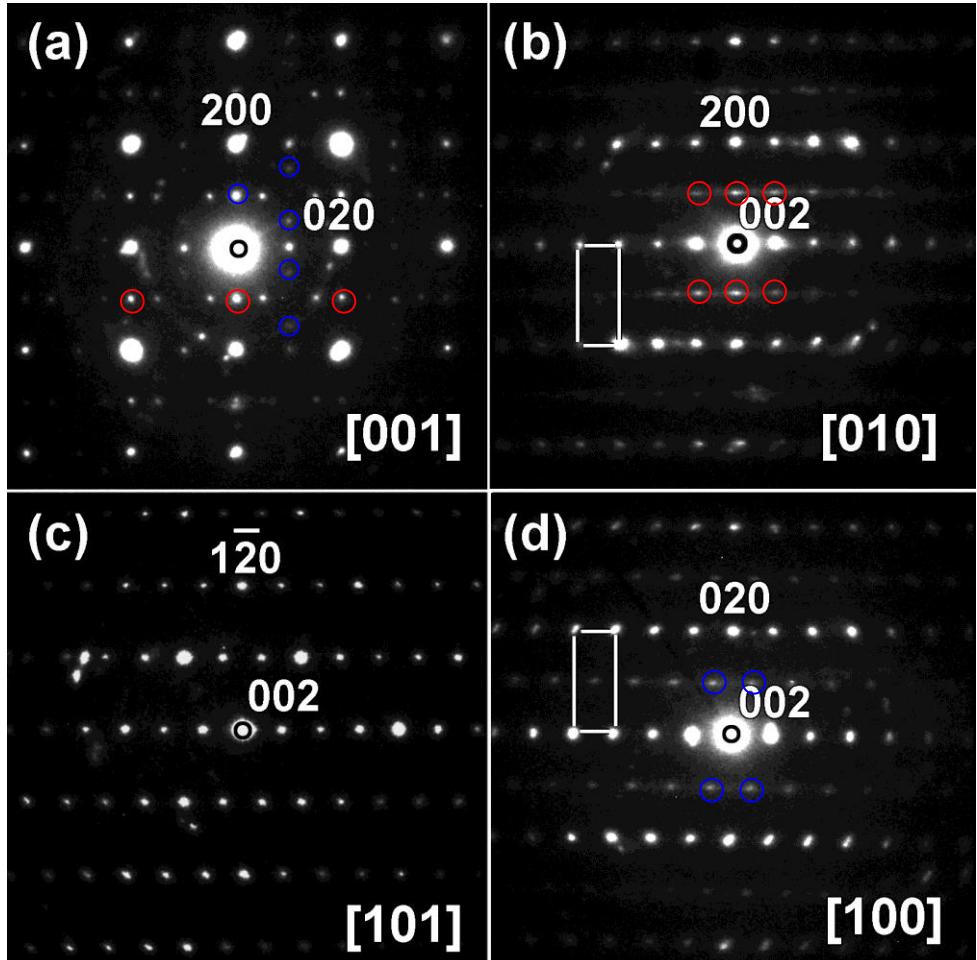
This structure is very similar to the AFe_2As_2 stoichiometric compounds :

$sp.gr. I4/mmm$
 $a \approx 4 \text{ \AA}$
 $c \approx 14 \text{ \AA}$

Legend:

- - Fe
- - As

Superstructures II and III with the ordered Fe and Rb vacancies



$$a_{II} = a_s + b_s$$

$$b_{II} = 2(a_s - b_s)$$

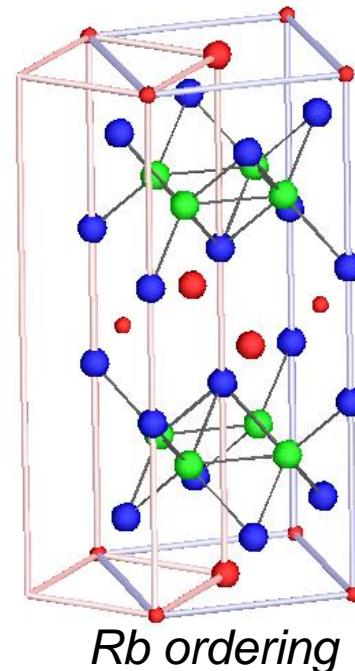
$$c_{II} = c_s$$

Space group I222

$$a_{II} = a_s \sqrt{2} \approx 5.5 \text{ \AA}$$

$$b_{II} = 2a_s \sqrt{2} \approx 11 \text{ \AA}$$

$$c_{II} = 14.5 \text{ \AA}$$



$$a_{III} = a_s + b_s$$

$$b_{III} = a_s - b_s$$

$$c_{III} = c_s$$

Space group Ammm

$$a_{III} = a_s \sqrt{2} \approx 5.5 \text{ \AA}$$

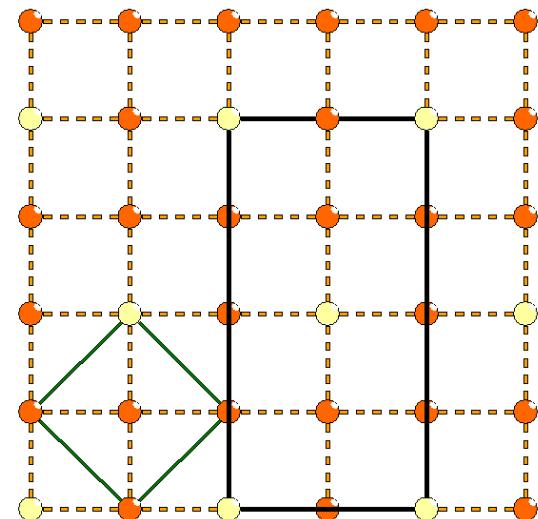
$$b_{III} = a_s \sqrt{2} \approx 5.5 \text{ \AA}$$

$$c_{III} = 14.5 \text{ \AA}$$

Different possibility for vacancy ordering for $A_1Fe_{1.5}Se_2$

$\sqrt{2} \times 2\sqrt{2}$ *Ibam*

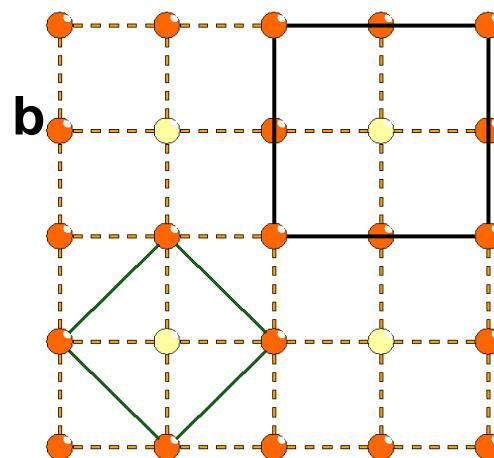
$a' = \sqrt{2}a$, $b' = 2\sqrt{2}a$, $c' = c$



Реализуется в $Tl_2Fe_3S_4$
[PRL 106, 087005 (2011)]

$\sqrt{2} \times \sqrt{2}$

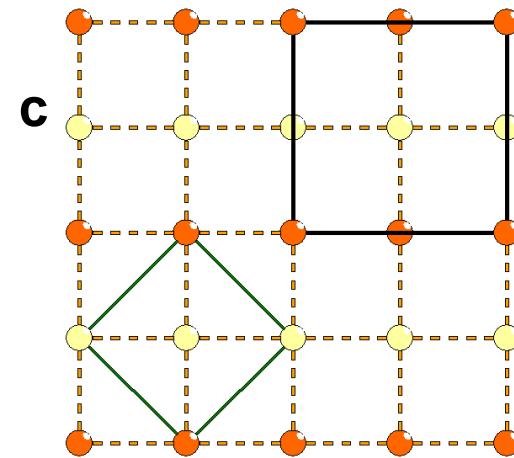
$a' = b' = \sqrt{2}a$, $c' = c$



Предсказан для $A_2Fe_3Se_4$

$\sqrt{2} \times \sqrt{2}$ *Pnma*

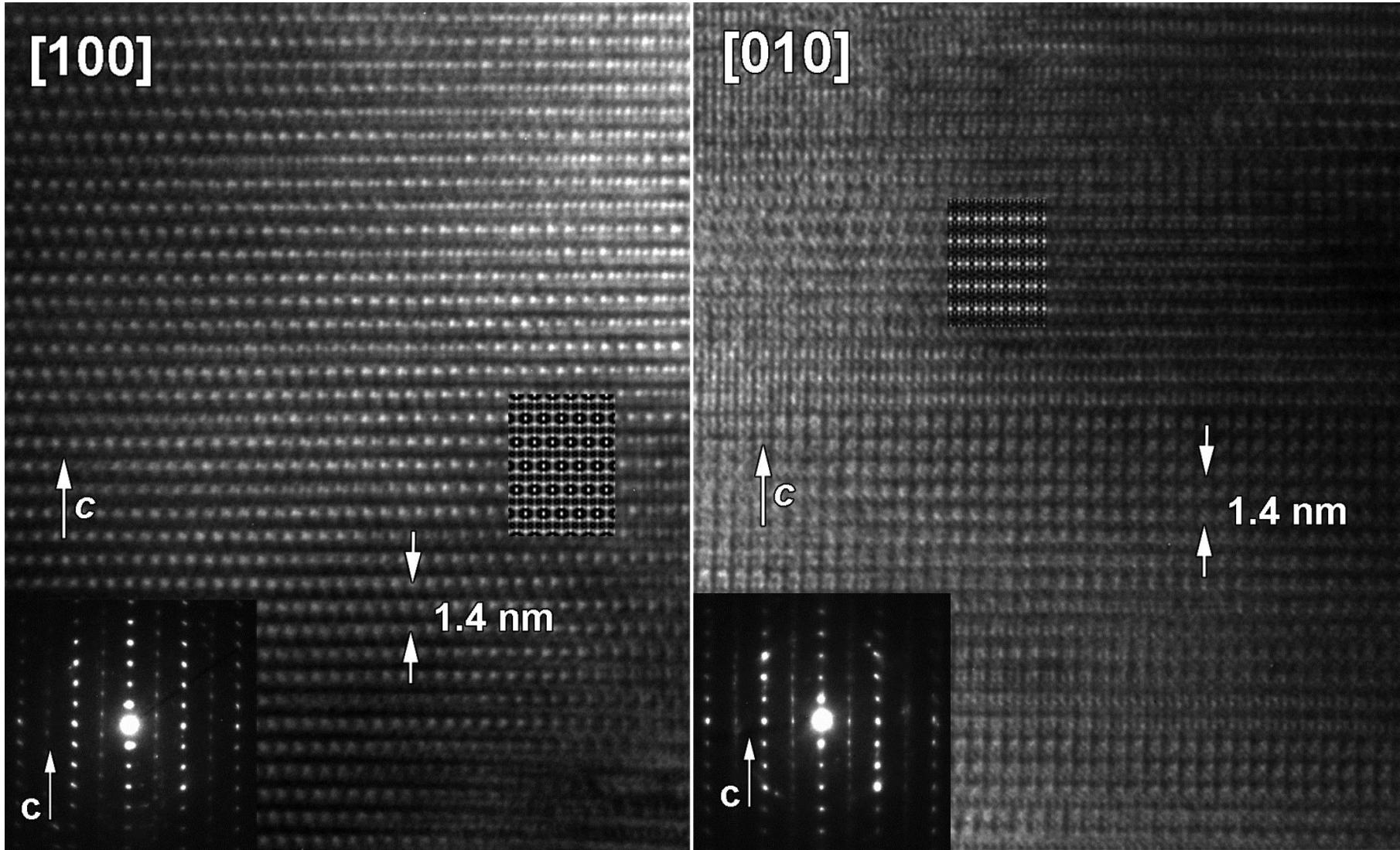
$a' \approx \sqrt{2}a$, $b' \approx \sqrt{2}a$, $c' = c$



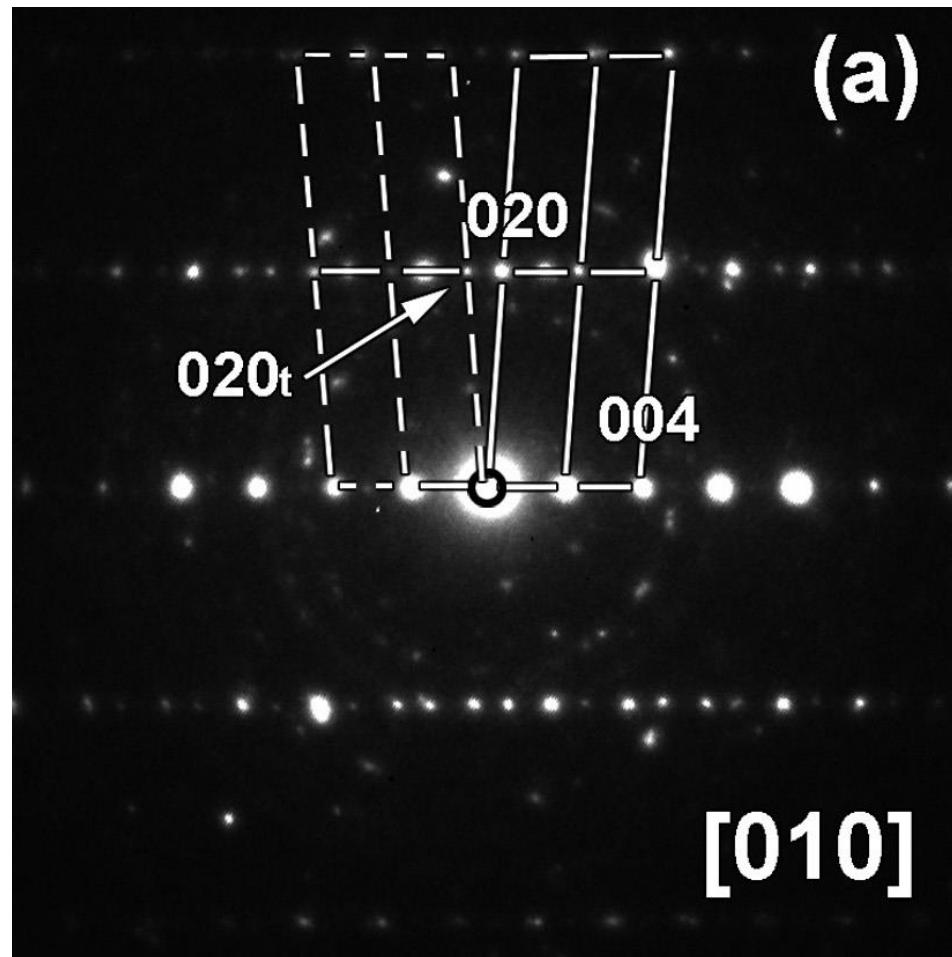
Реализуется в $KFe_{1.5}Se_2$
[arXiv:1102.3674v1].

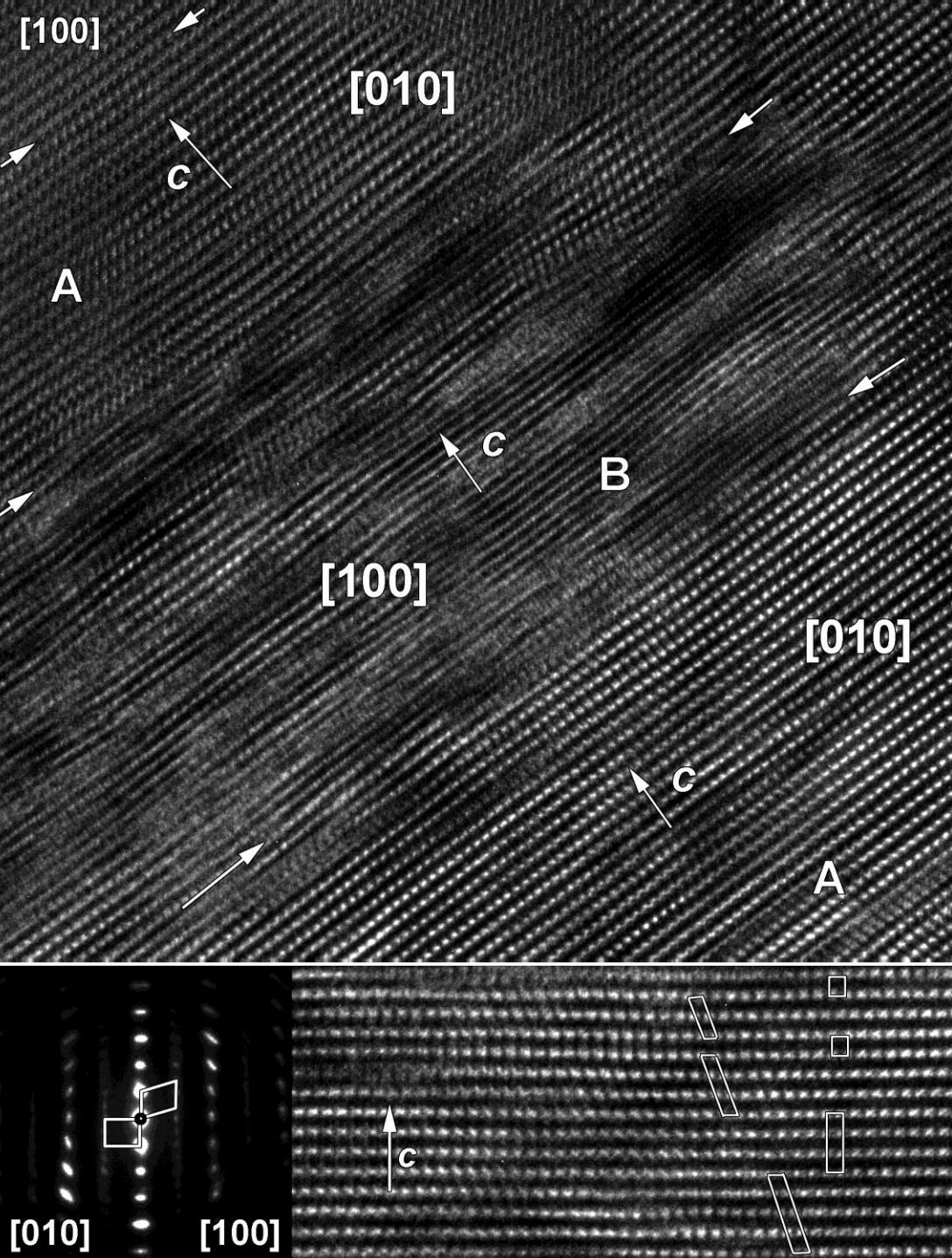
HRTEM study: SC sample

Structure model based on **Superstructure III** with Ammm space group and cell parameters $a = b \approx 5.5 \text{ \AA}$, $c = 14.5 \text{ \AA}$



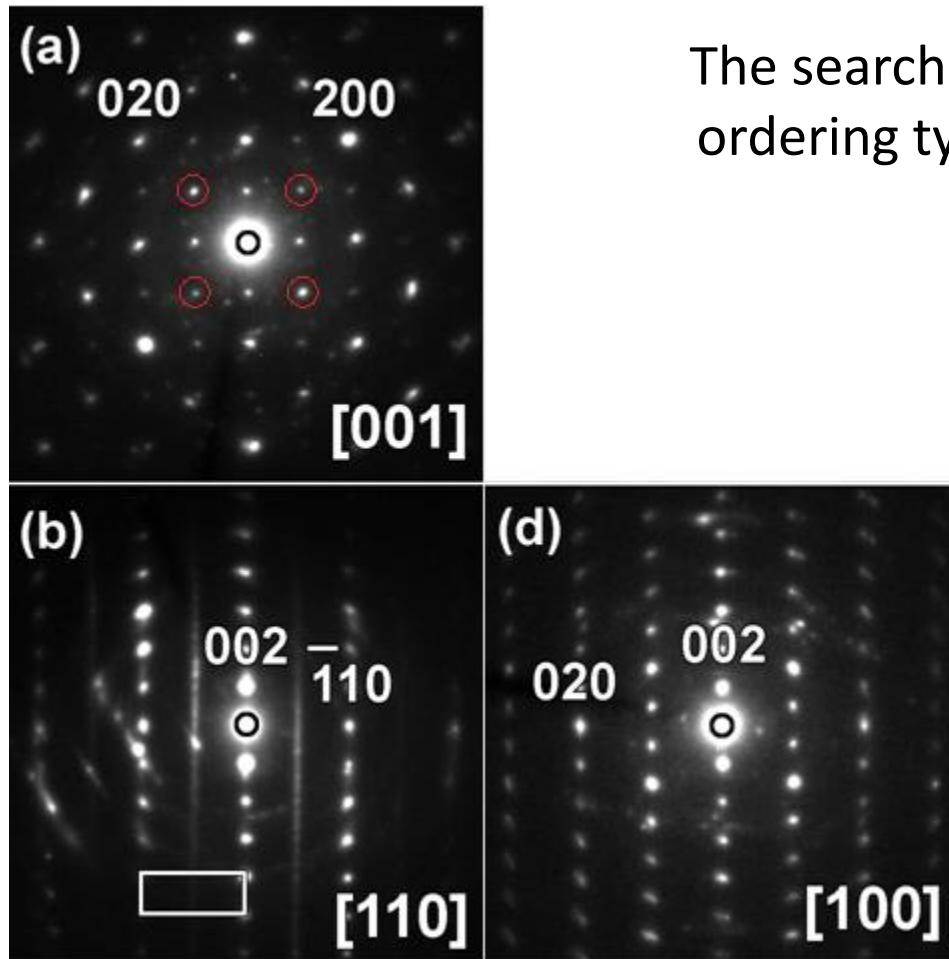
The monoclinic distortion in the SC-sample



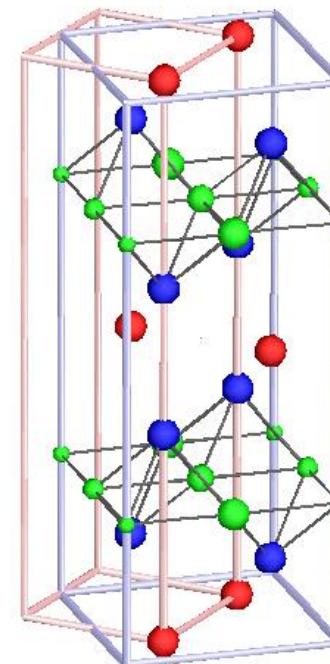


HRTEM study:
Non-SC sample

The structure features of non-SC $\text{Rb}_{0.75(1)}\text{Fe}_{1.66(3)}\text{Se}_2$



The search of the alternative Fe vacancy ordering types among the subgroups of I4/mmm



Pccm

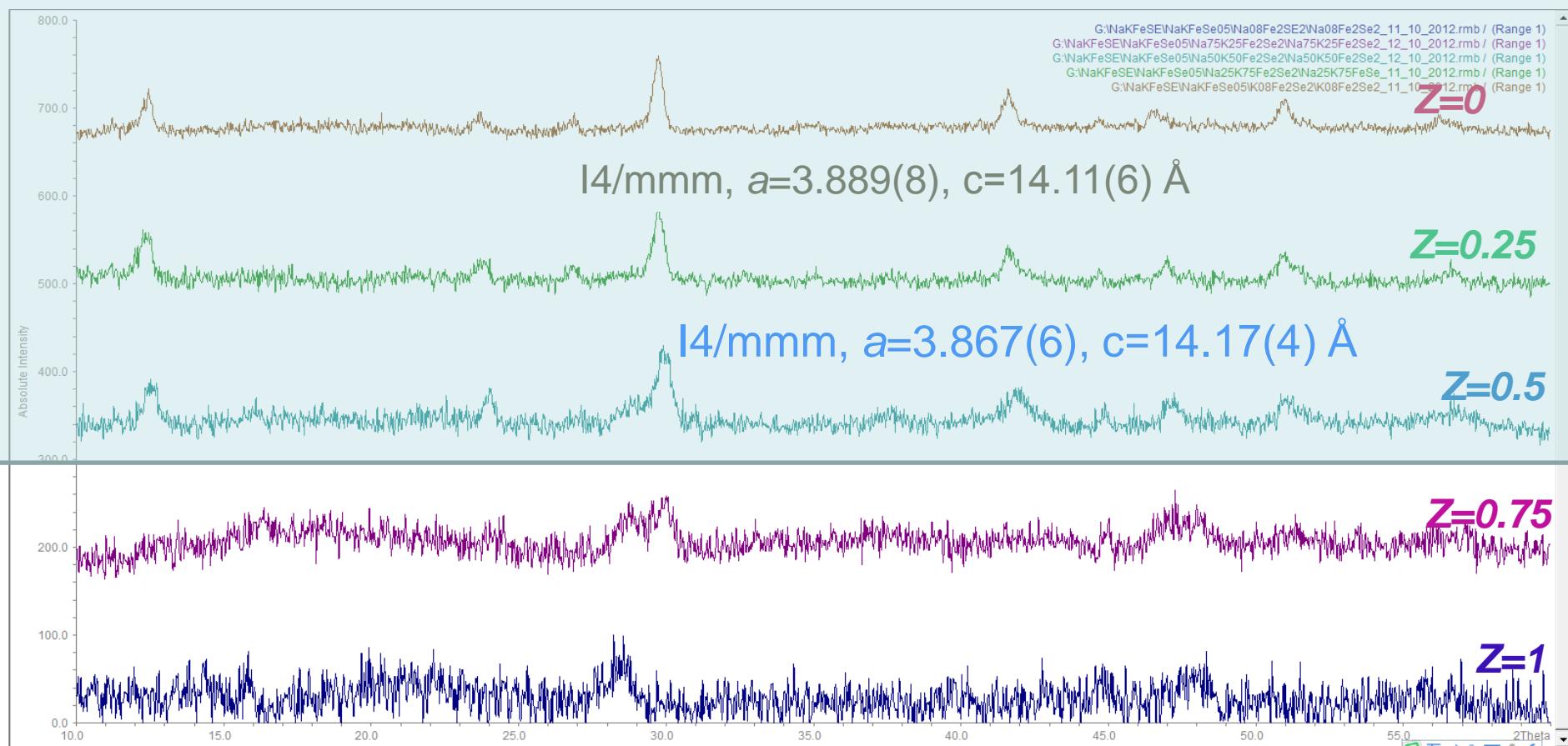
$$a=b=\sqrt{2}a_s=5.5\text{\AA}$$
$$c=14.5\text{\AA}$$

Conclusions

1. For the first time the superconducting and the nonsuperconducting RbxFe2-ySe2 materials with close composition were investigated in detail by ED and TEM.
2. It is shown that the alternation of ordered and disordered regions is characteristic for both SC and non-SC materials.
3. Three types of electron diffraction patterns were found for the superconducting RbxFe2-ySe2 sample, of which one is observed for the first time and originates from alkali metal ordering.
4. Moreover, for the superconducting RbxFe2-ySe2 material a monoclinic distortion with $\beta \sim 87^\circ$ was observed. This monoclinic distortion seems to be an attribute of the superconducting material only.
5. The non-superconducting sample the orthogonality of the crystallographic axes is preserved.

III. Synthesis and study of the properties of ferroselenids $(\text{Na},\text{K})_x\text{Fe}_{2-y}\text{Se}_2$ with partial substitution of K by Na.

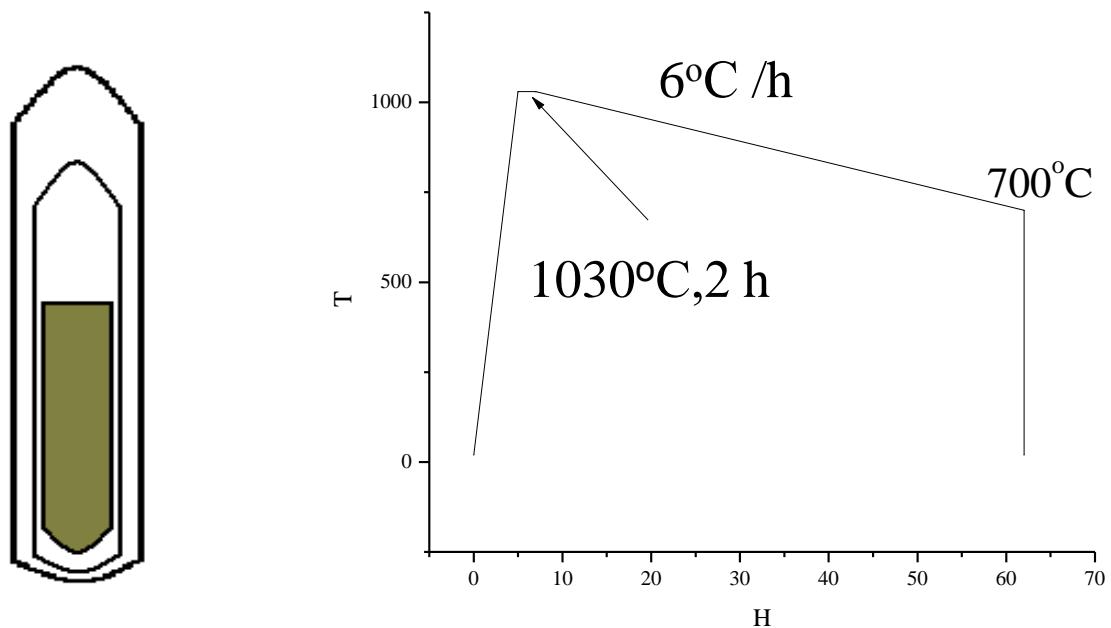
Synthesis of polycrystalline $(\text{Na}_z\text{K}_{1-z})_x\text{Fe}_{2-y}\text{Se}_2$



Synthesis of Single crystalline samples $(\text{Na}_z\text{K}_{1-z})_x\text{Fe}_{2-y}\text{Se}_2$

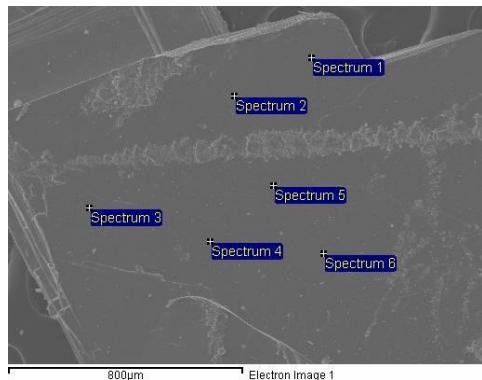
1. Prereaction $0.8 \text{ A} + 2\text{FeSe} \longrightarrow \langle\langle \text{A}_{0.8}\text{Fe}_2\text{Se}_2 \rangle\rangle \text{ A= K, Na}$ (380°C 6 h)

2)

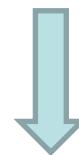
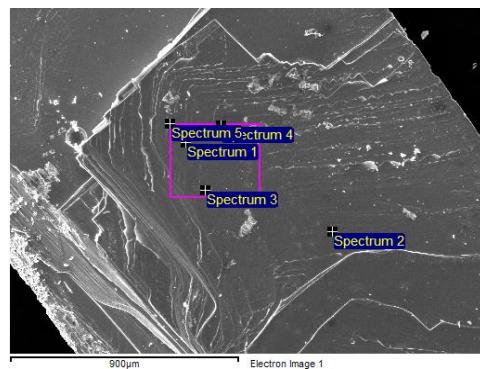


EDX

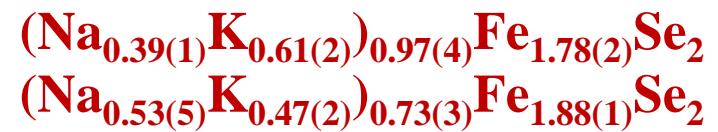
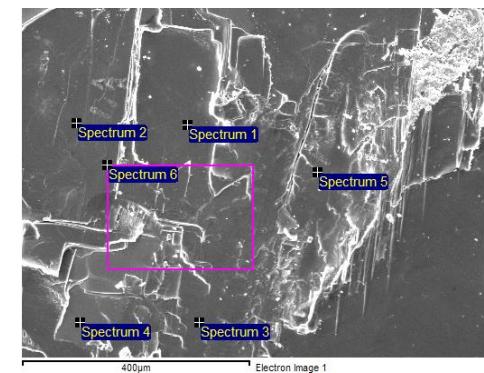
Z=0



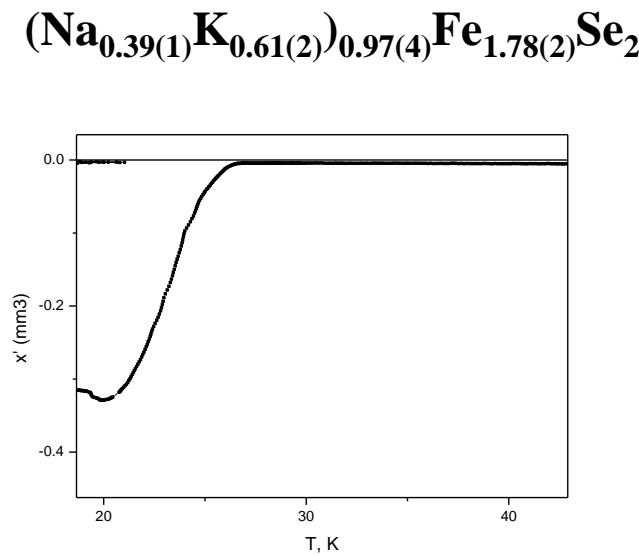
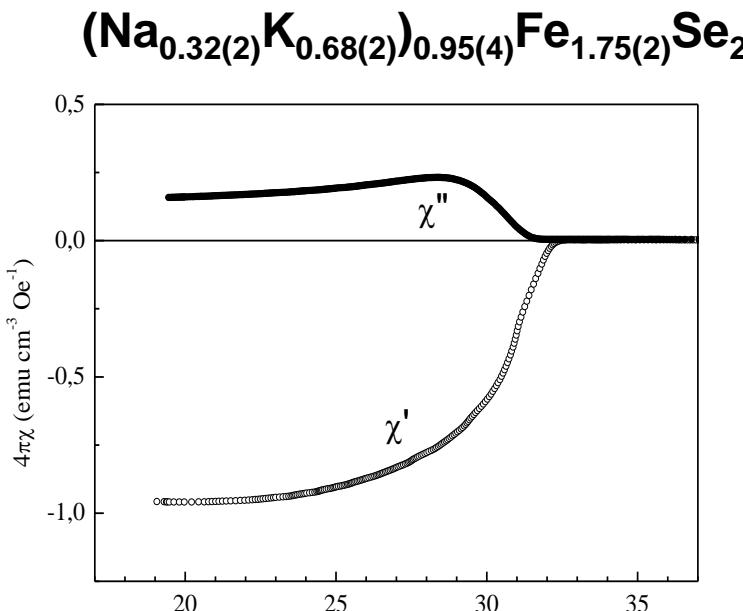
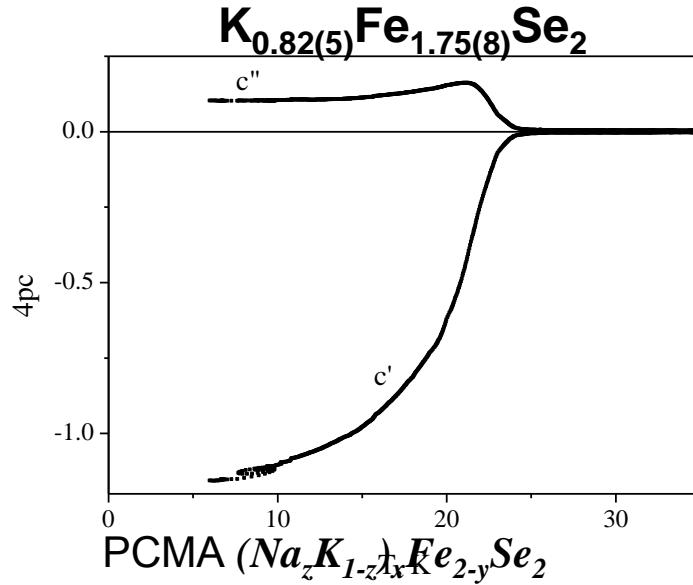
Z=0.3



Z=0.4



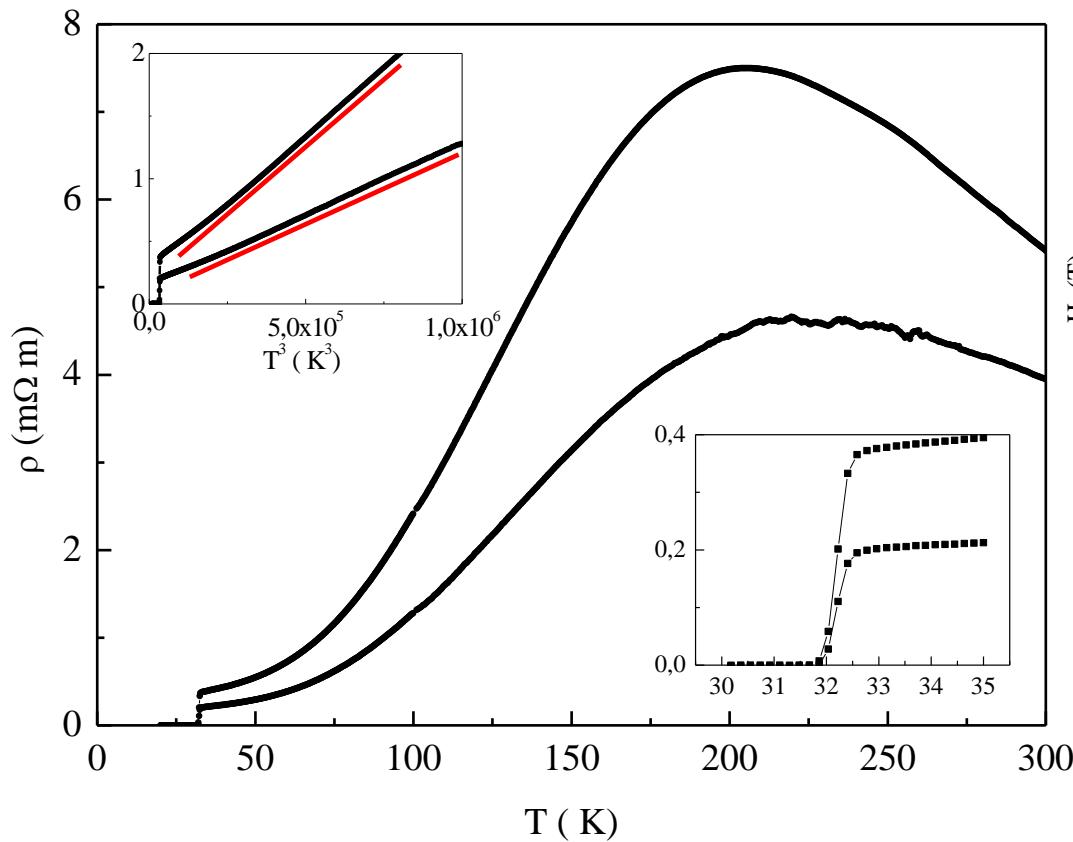
Magnetization measurements of $K_xFe_{2-y}Se_2$ and $(K,Na)_xFe_{2-y}Se_2$



X-ray diffraction analisys $(\text{Na}_z\text{K}_{1-z})_x\text{Fe}_{2-y}\text{Se}_2$

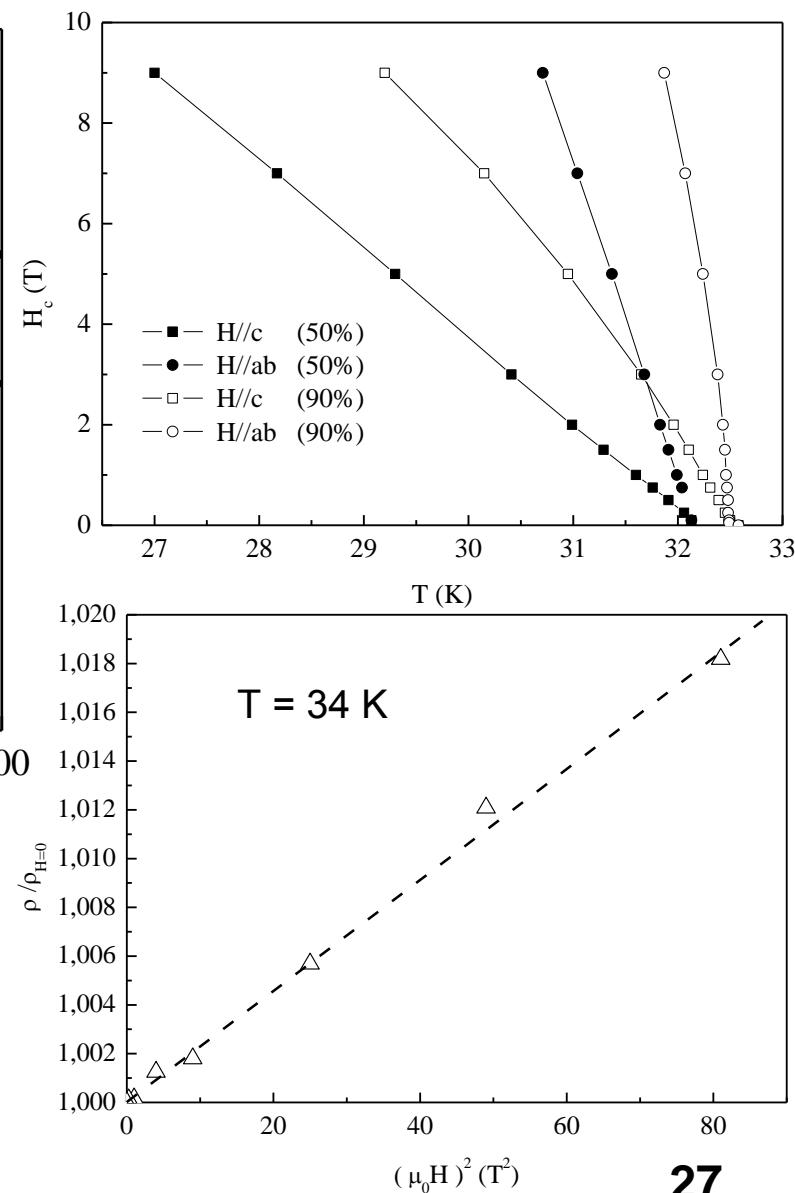
Z	Composition	Параметры Å		T_c K
		a	c	
0*	$\text{Cs}_x\text{Fe}_{2-y}\text{Se}_2$	8,8582	15,2873	28.5
0**	$\text{Rb}_x\text{Fe}_{2-y}\text{Se}_2$	8,799	14,576	31.5
0**	$\text{K}_{0.82(5)}\text{Fe}_{1.75(8)}\text{Se}_2$	8,73	14,115	29.5
0	$\text{K}_{0.82(5)}\text{Fe}_{0.75(8)}\text{Se}_2$	8.71(1)	14.11(2)	24
0.25***	$(\text{Na}_{0.21(4)}\text{K}_{0.79(2)})_{0.67(2)}\text{Fe}_{1.49(1)}\text{Se}_2$	8.67(2)	14.23(5)	31
0.3	$(\text{Na}_{0.32(2)}\text{K}_{0.68(2)})_{0.95(4)}\text{Fe}_{1.75(2)}\text{Se}_2$	8.72(2)	14.16(8)	32.4
0.4	$(\text{Na}_{0.39(1)}\text{K}_{0.61(2)})_{0.97(4)}\text{Fe}_{1.78(2)}\text{Se}_2$ $(\text{Na}_{0.53(5)}\text{K}_{0.47(2)})_{0.73(3)}\text{Fe}_{1.88(1)}\text{Se}_2$	8.725(6)	14.23(4)	26

Resistivity measurements of $(\text{Na}_{0.3}\text{K}_{0.7})_x\text{Fe}_{2-y}\text{Se}_2$

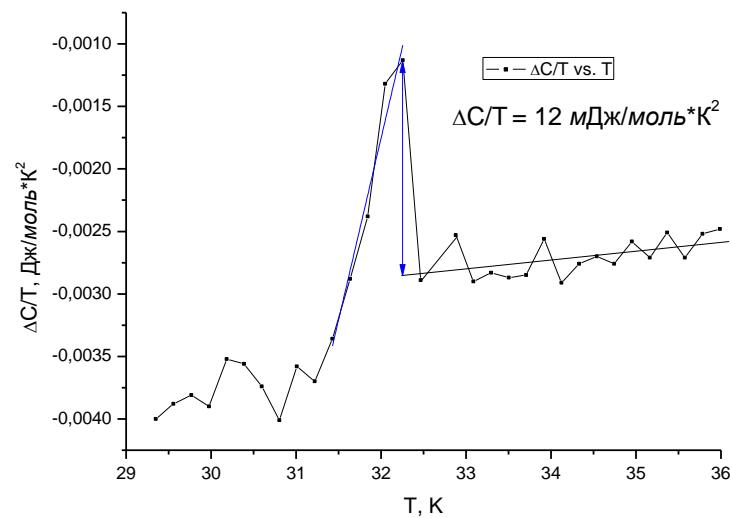
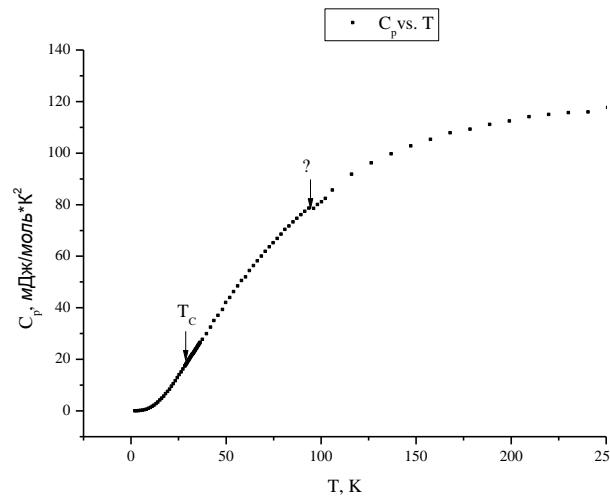
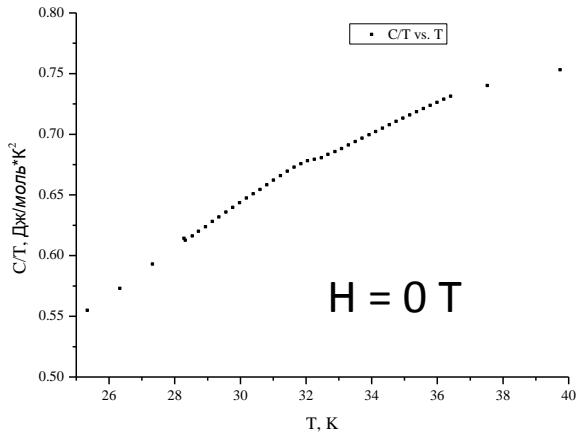


$$\begin{aligned}-dH_{c2}^c/dT &= 24.8 \text{ T/K} \\ -dH_{ab,c2}^c/dT &= 3.4 \text{ T/K}\end{aligned}$$

positive transverse magnetoresistance $\sim H^{1/2}$

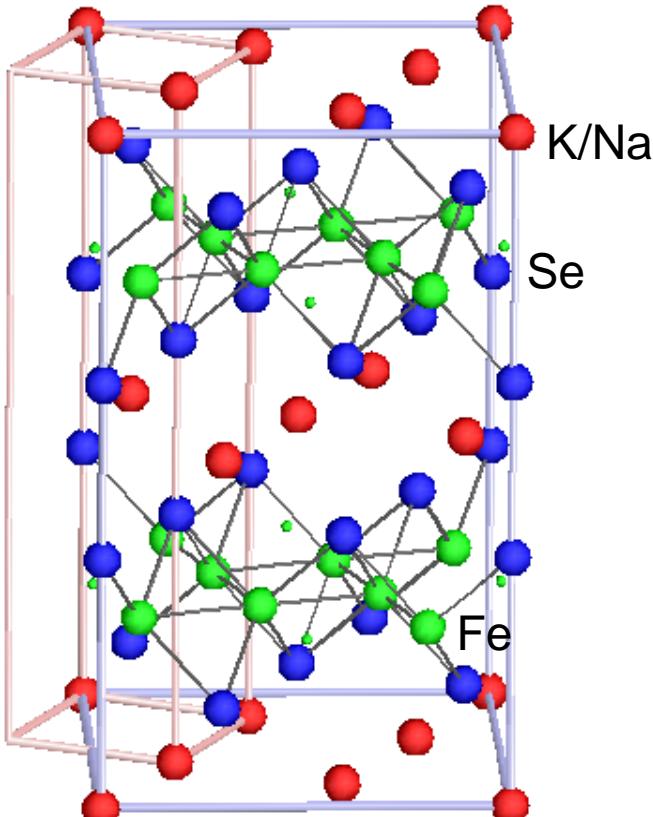


Specific heat measurements of $(\text{Na},\text{K})_x\text{Fe}_{2-y}\text{Se}_2$



- There are 2 features on the specific heat curve: at $T = 32 \text{ K}$ (corresponds T_C), and at $T = 95 \text{ K}$.
 - The jump of C/T at $T=T_C$ is found to be $\sim 12 \text{ mJ/mol}^{\circ}\text{K}^2$.
 - In the low temperature range the data can be fitted with the equation:
- $$C_p(T) = \gamma T + \beta T^3$$
- where γT – electron contribution, βT^3 – phonon contribution.
- At $H=0 \text{ T}$ is found to be $\sim 0.4 \text{ mJ/mol}^{\circ}\text{K}^2$

High-resolution X-ray diffraction study of $K_xFe_{2-y}Se_2$ and $(K_{0.7}Na_{0.3})_xFe_{2-y}Se_2$ using SR



Features:

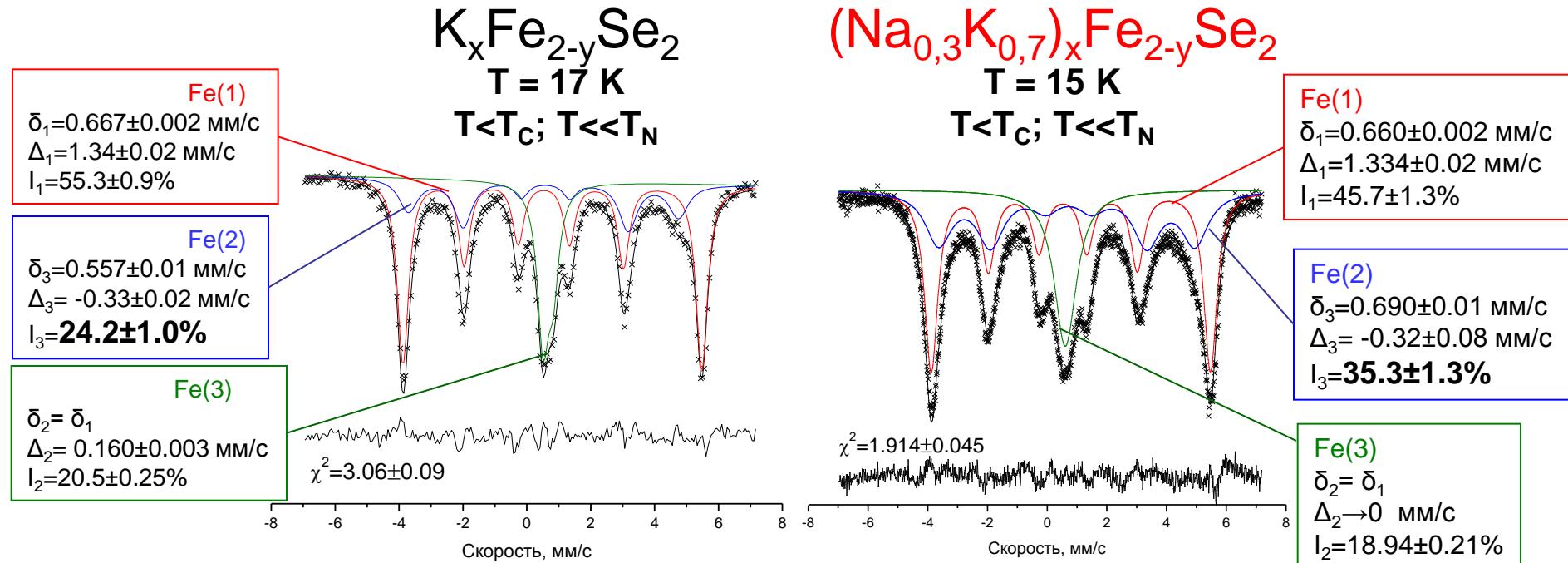
Poor crystallinity,
multiple phases →
Significant broadening
of the reflections

HRXRD revealed the presence of at least two phases: $A_2Fe_4Se_5$ (s.g. I4/m) и $A_xFe_2Se_2$ (s.g. I4/mmm). The refinement of the cell parameters led to the following values:

	$A_2Fe_4Se_5$ S.g. I4/m	$A_xFe_2Se_2$ S.g. I4/mmm
$A=K$	$a = 8.7031(2)\text{\AA}$ $c = 14.1588(4)\text{ \AA}$	$a = 3.8467(3)\text{ \AA}$ $c = 14.2312(4)\text{ \AA}$
$A=K,Na$	$a = 8.6870(3)\text{ \AA}$ $c = 14.1148(5)\text{ \AA}$	$a = 3.9397(3)\text{ \AA}$ $c = 14.2241(2)\text{ \AA}$

Sodium doping significantly increases the a parameter of the $A_xFe_2Se_2$ phase whereas the c parameter remains essentially the same. For the $A_2Fe_4Se_5$ phase the opposite tendency is observed
The incorporation of Na induces a strong disorder

The ^{57}Fe Mössbauer spectra of $(\text{Na}_{0.3}\text{K}_{0.7})_x\text{Fe}_{2-y}\text{Se}_2$ and $\text{K}_x\text{Fe}_{2-y}\text{Se}_2$ polycrystalline absorbers



- ^{57}Fe Mössbauer spectrum in the temperature range below T_N can be fitted as a superposition of two magnetic sextets Fe1 and Fe2 and one weakly spitted quadrupole doublet Fe3.
- The co-existence of two magnetically ordered phases with magnetic moments on the Fe1 and Fe2 atoms being 2 and $2.2 \mu_B/\text{Fe}$, respectively. It was shown that the magnetic moments of the Fe1 are aligned along the c-axis while the magnetic moments of the Fe2 are lying in the ab-plane.

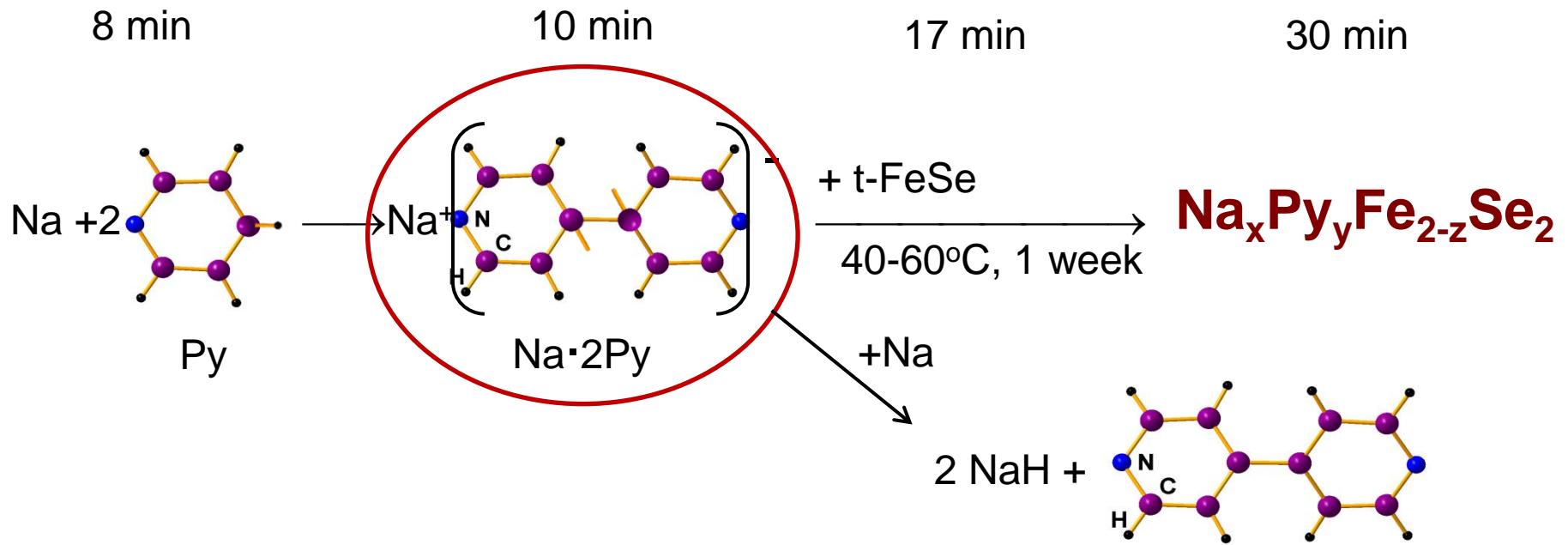
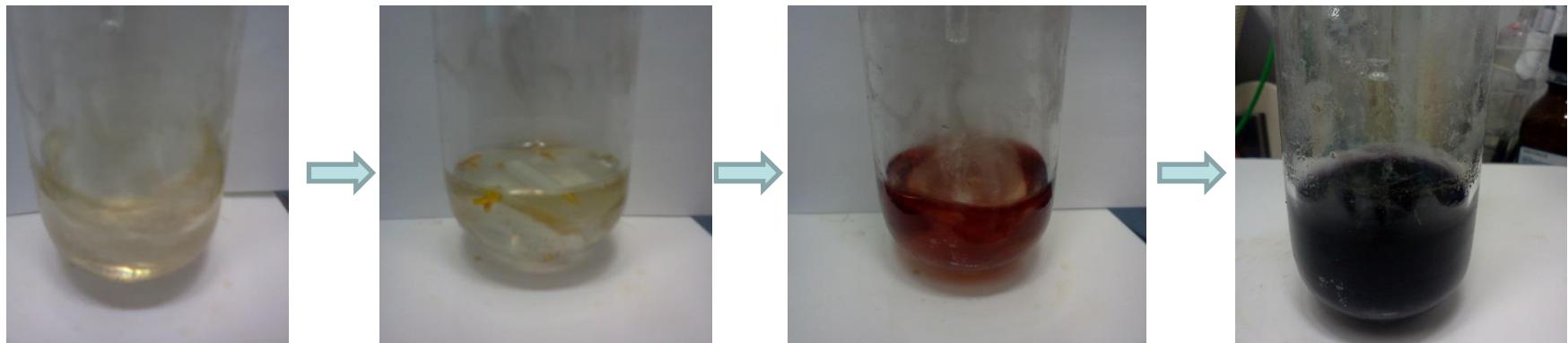
Conclusions

1. Samples $(\text{Na}_z\text{K}_{1-z})_x\text{Fe}_{2-y}\text{Se}_2$ with different degree of substitution was synthesized for the first time. The composition and microstructure of the samples were studied by X-ray diffraction and EDX. It is shown that the maximum degree of substitution of potassium for sodium is about 40%.
2. The study of magnetic and transport properties showed that the sodium-potassium ferroselenids are superconductors with maximum T_C 32.5 K.
3. The study of the microstructure of the sample $(\text{Na}_{0.3}\text{K}_{0.7})_x\text{Fe}_{2-y}\text{Se}_2$ by synchrotron radiation and Mössbauer spectroscopy revealed a phase separation of the sample. Mössbauer spectra consist of two magnetic and one paramagnetic components and, when grinding the content of second magnetic component increases from 6% to 35%, while the superconducting fraction decreases sharply.

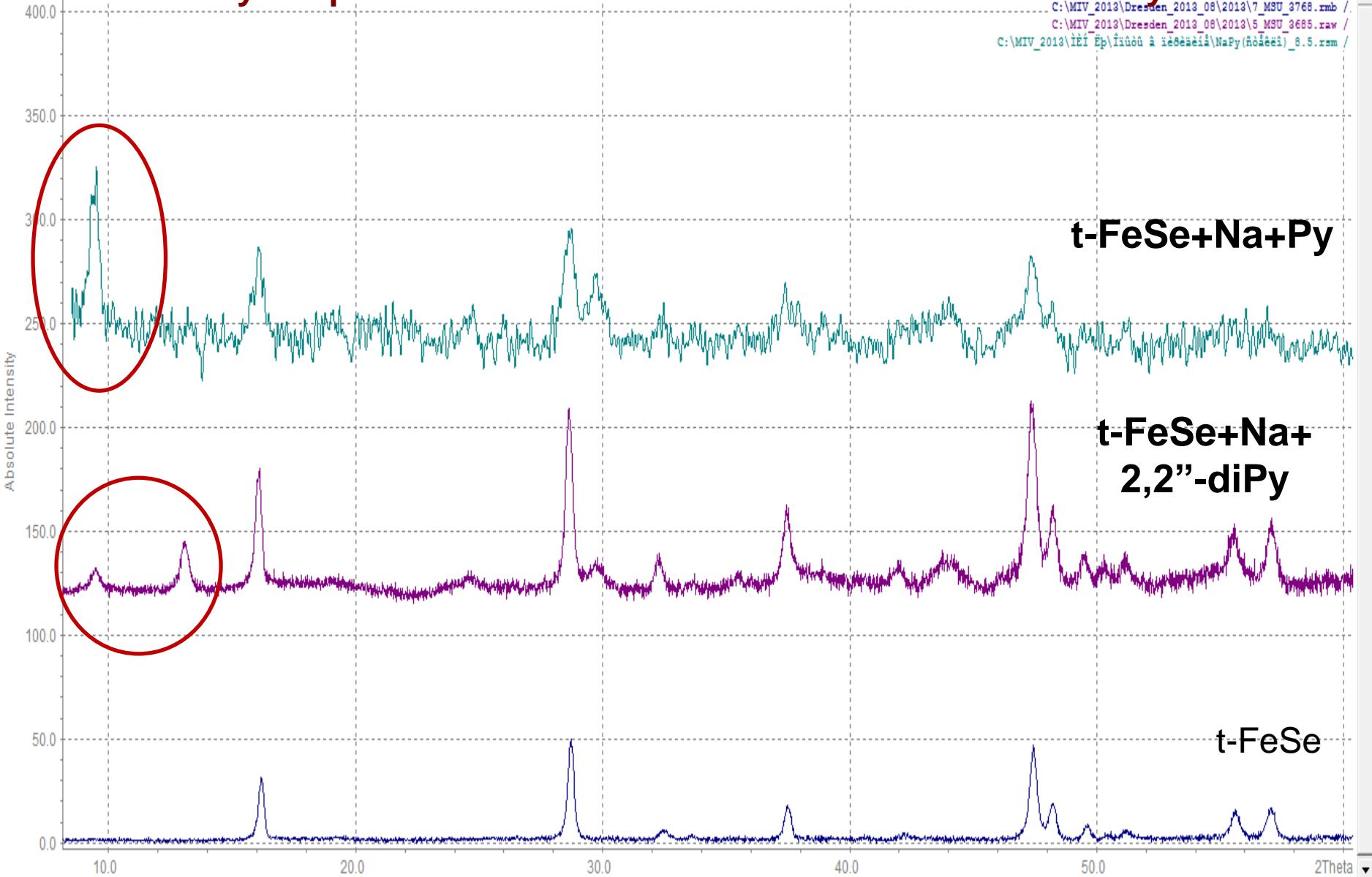
IV. Synthesis of ferroselenids $A_x(C_5H_5N)_yFe_{2-z}Se_2$
(A=Li, Na) by intercalation of Li and Na to FeSe in pyridine.

Synthesis of samples $A_x(C_5H_5N)_yFe_{2-z}Se_2$ ($A=Li, Na$) by methods a soft chemistry

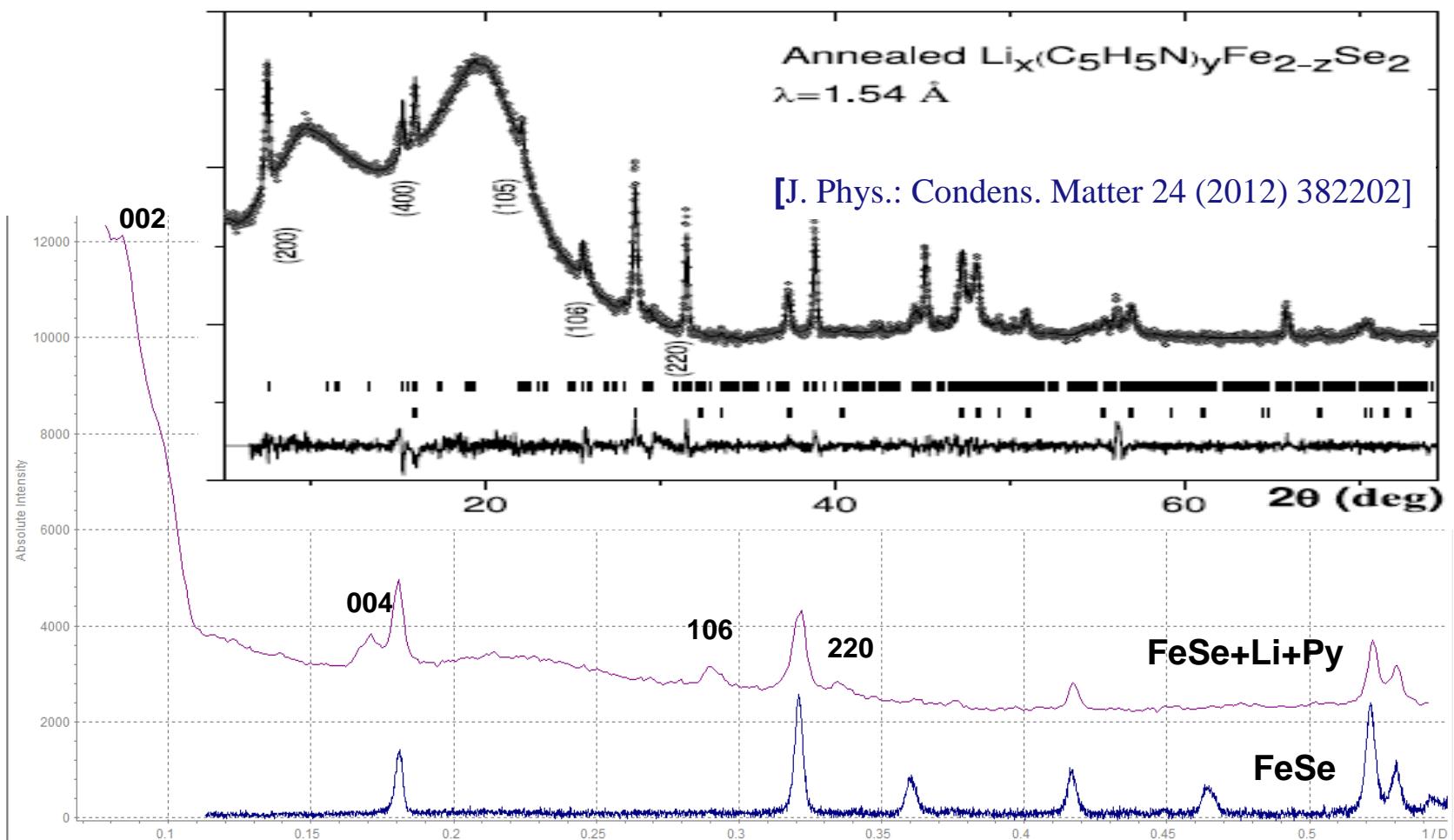
The intercalation of alkali metal in t-FeSe from pyridine solution.



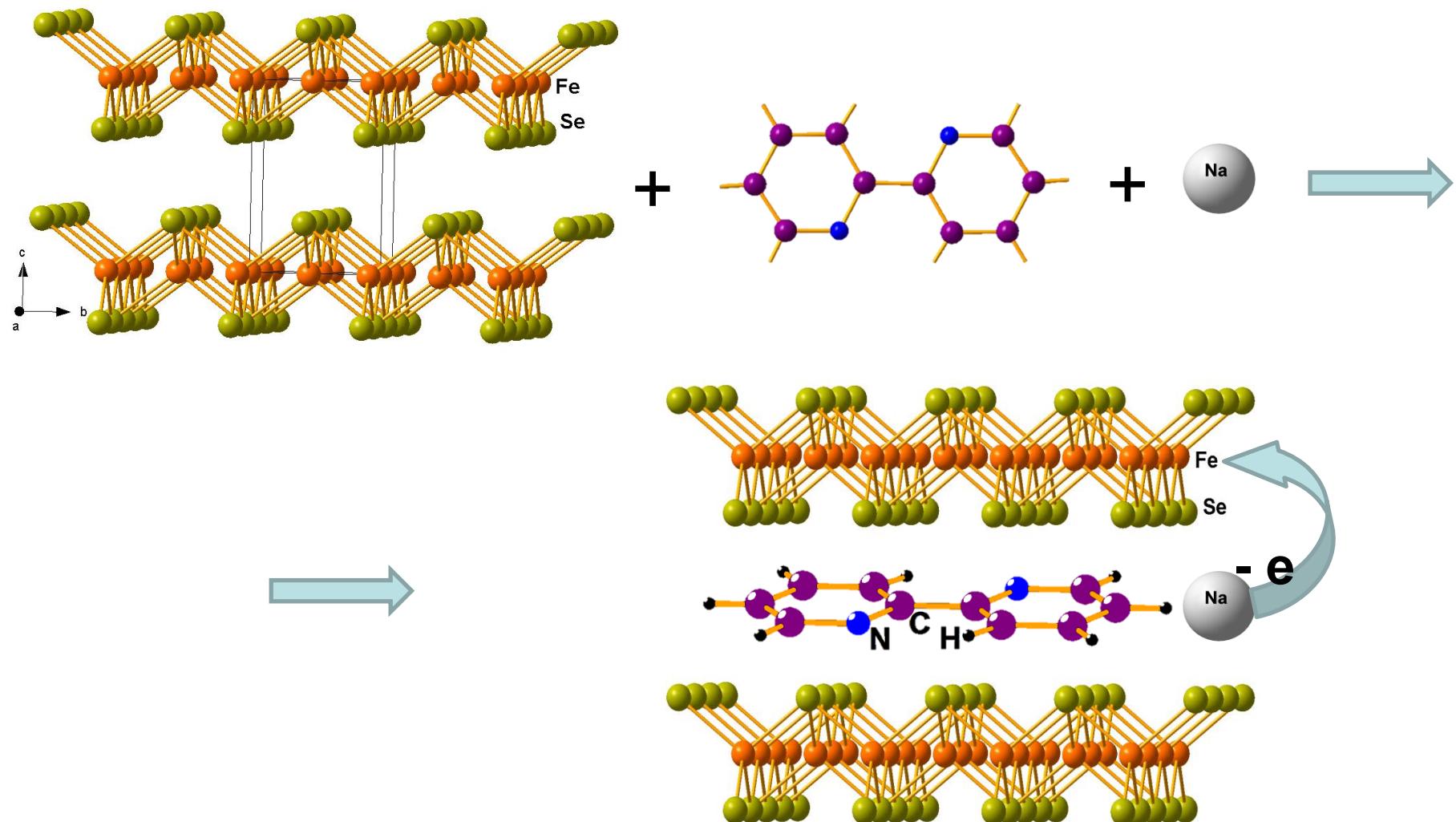
XRD analisys of products of reaction FeSe with Na solutions in Py and THF

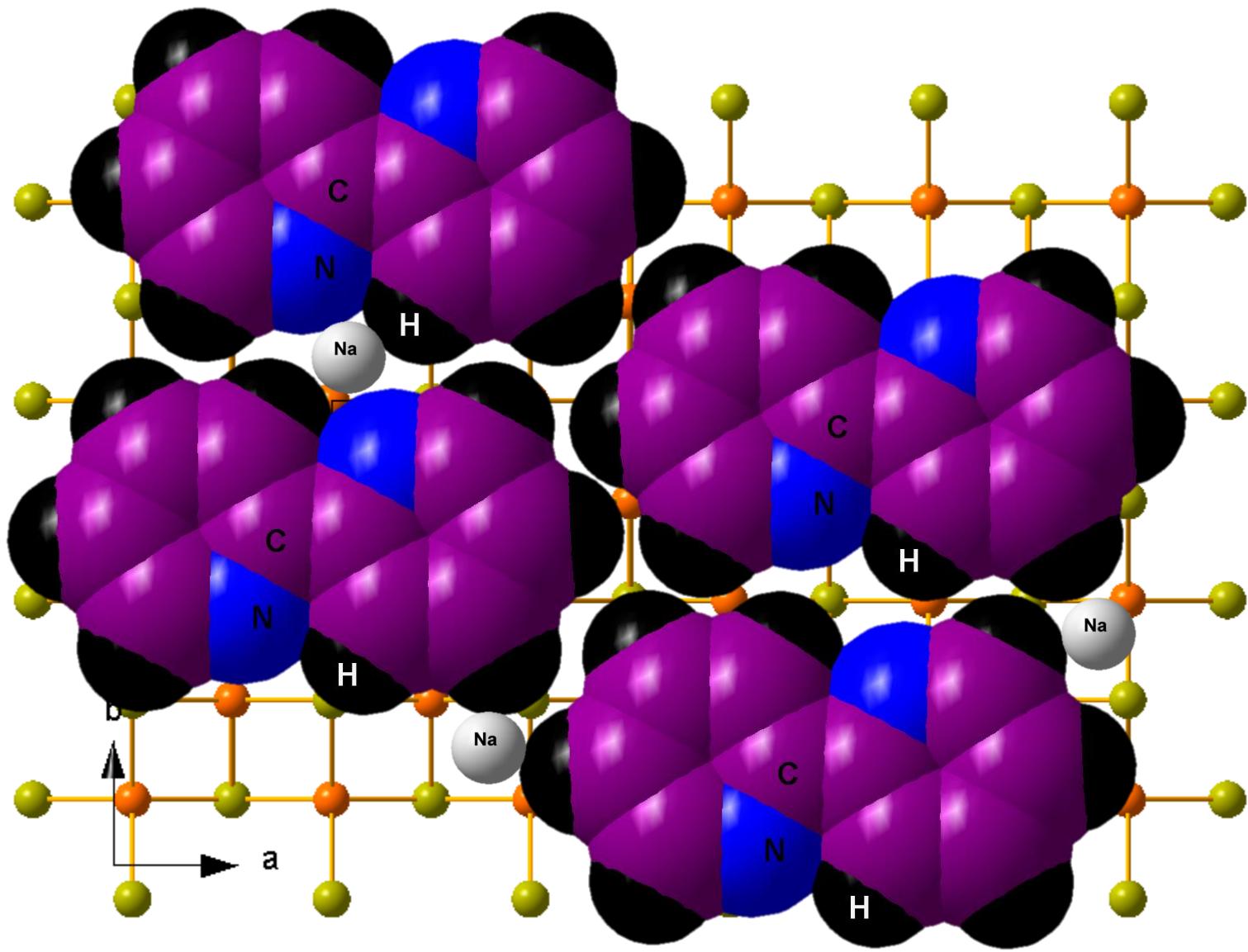


Li + FeSe + Py

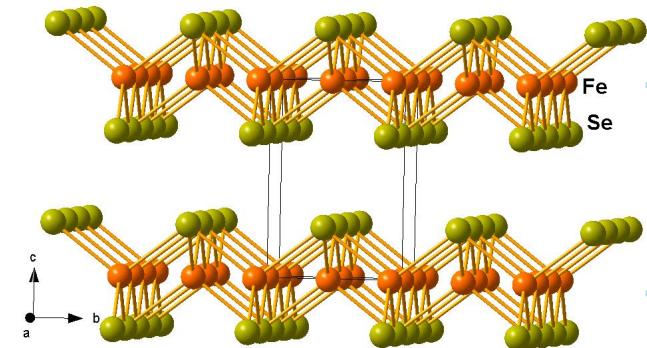


Intercalation of t-FeSe by sodium and 2,2'-dipyridyl in THF

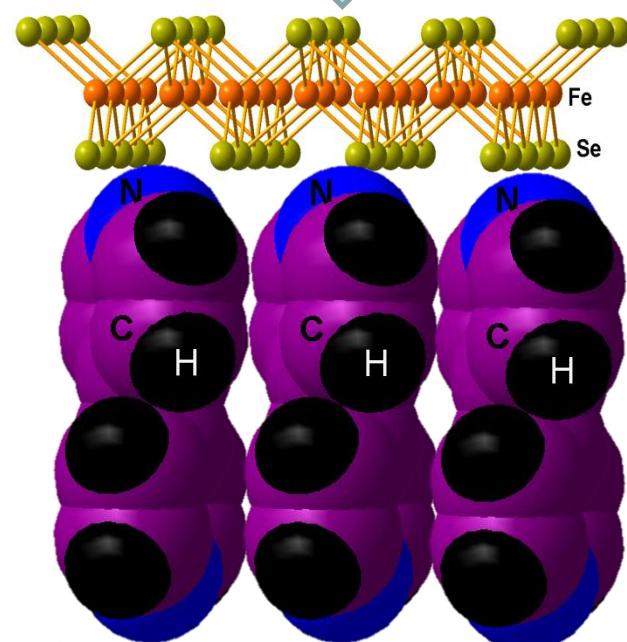




The relationship between the interlayer distance and the Van der Waals dimensions of the molecular-spacer



$\text{Li} + \text{Py}$

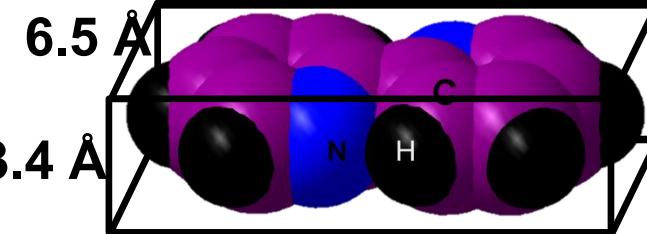


$\text{Li}_x\text{diPy}_y\text{Fe}_2\text{Se}_2$

5.52 Å

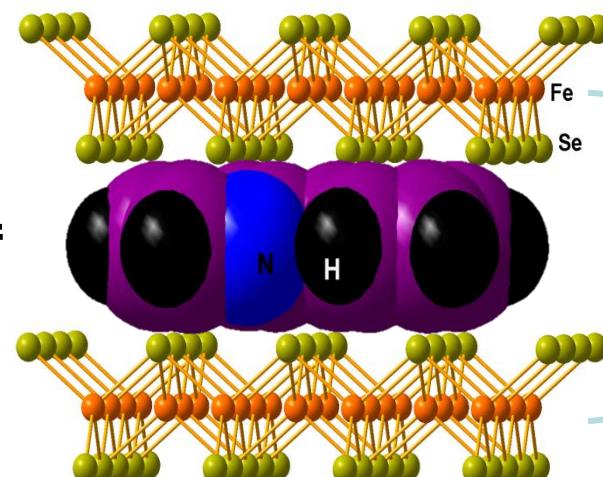
$\text{Na} + \text{diPy}$
 THF

Van der Waals dimensions of
2,2'-dipyridyl



11.3 Å (10.1 Å for 4,4'-diPy)

$$\begin{aligned}C &= 16.1 \approx \\&5.5 + 10.1 = \\&= 15.5 (\text{\AA})\end{aligned}$$



$\text{Na}_x\text{diPy}_y\text{Fe}_2\text{Se}_2$

$$\begin{aligned}C &= 9.34 \approx \\&5.5 + 3.4 = \\&= 8.9 (\text{\AA})\end{aligned}$$

Conclusions

1. Lithium and sodium ferroselenides was prepared by the methods of "soft chemistry" from the solution in pyridine (20-60 °C) by two ways:
 - by alkali metal intercalation in FeSe;
 - by substitution of potassium in $K_xFe_{2-y}Se_2$ on Na or Li.
2. It is shown that as spacer molecules can act dipyridyl.

Collaborations



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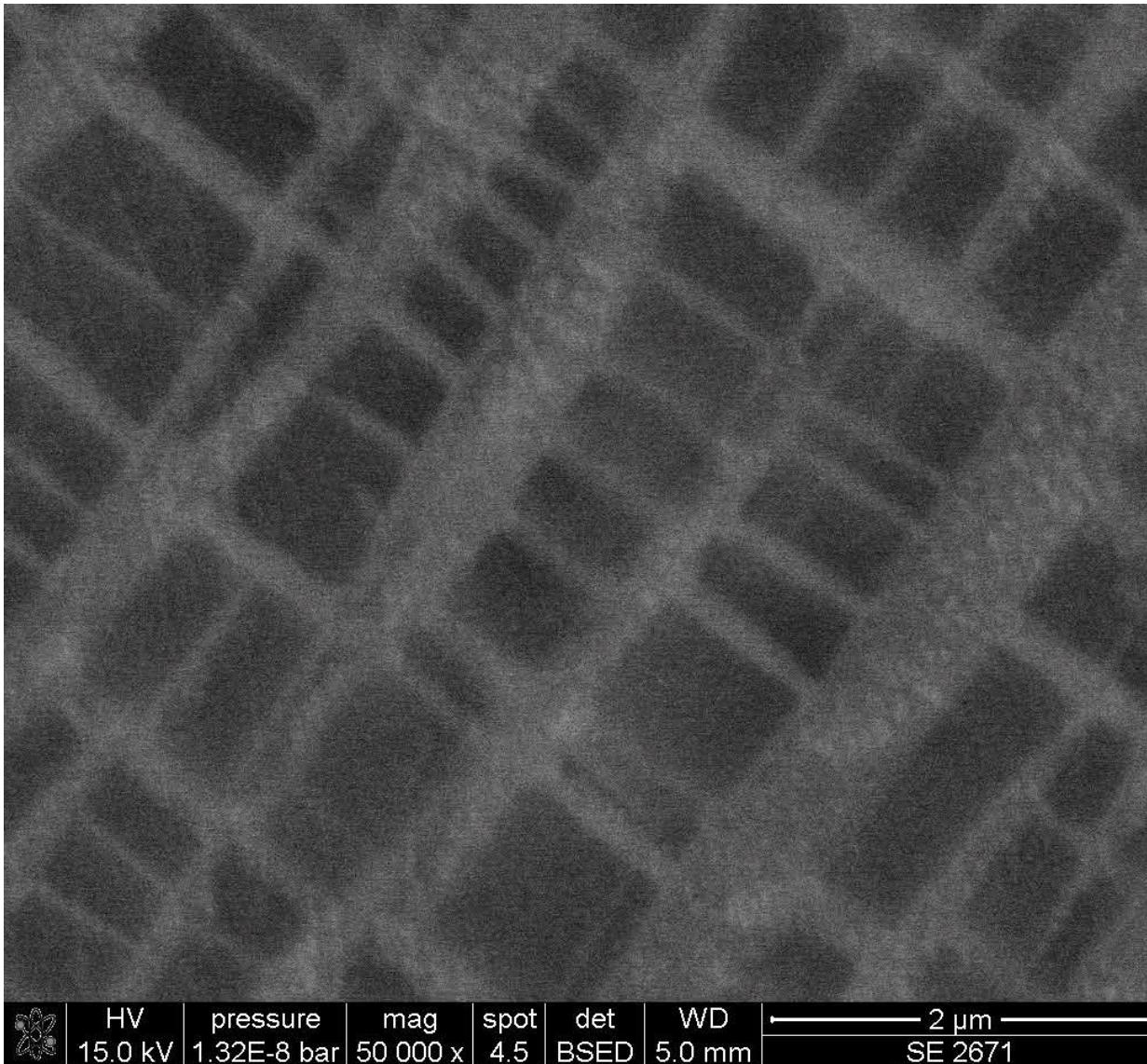


Lebedev O.I.

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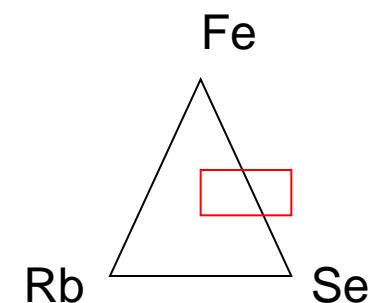
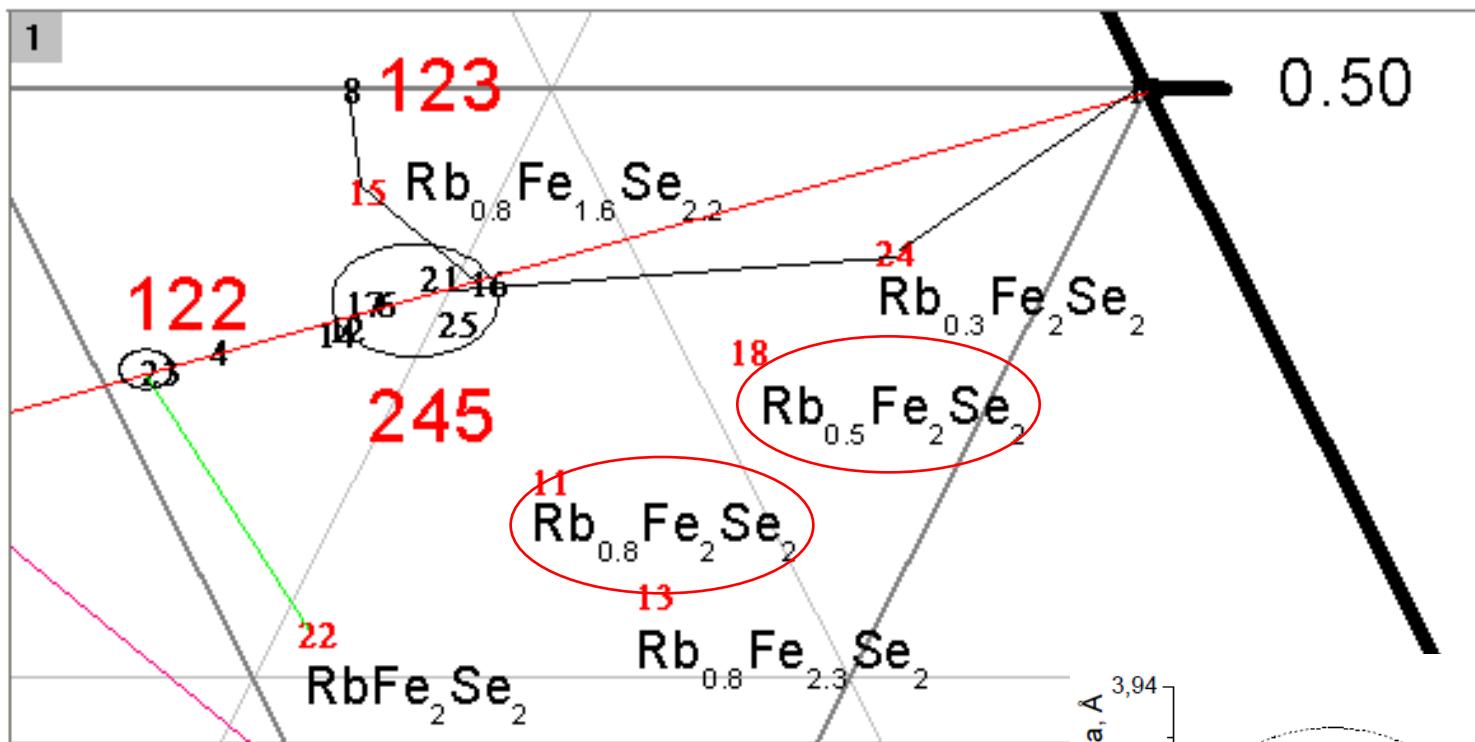
Thank you for attention!

Spinodal decomposition?



HV | pressure | mag | spot | det | WD | 2 μm
15.0 kV | 1.32E-8 bar | 50 000 x | 4.5 | BSED | 5.0 mm | SE 2671

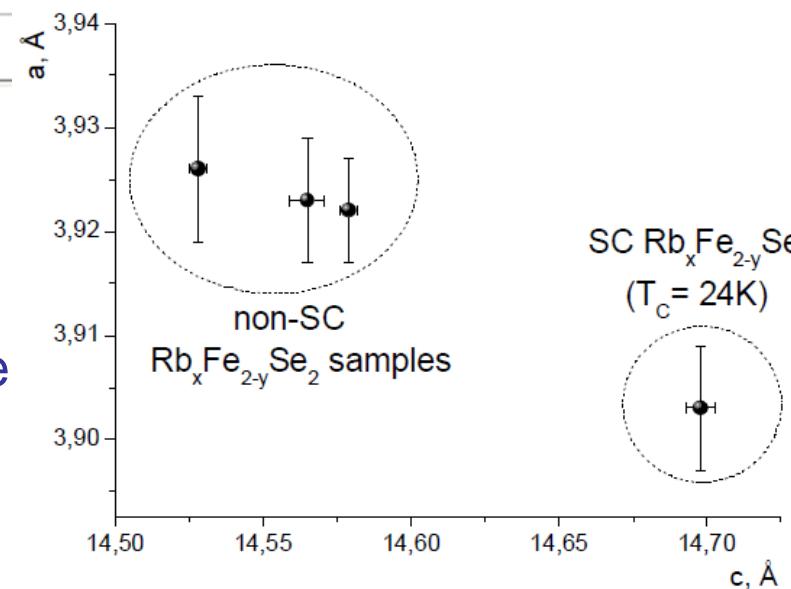
Compositional diagram Rb-Fe-Se



A different ratio Rb-Fe-Se:

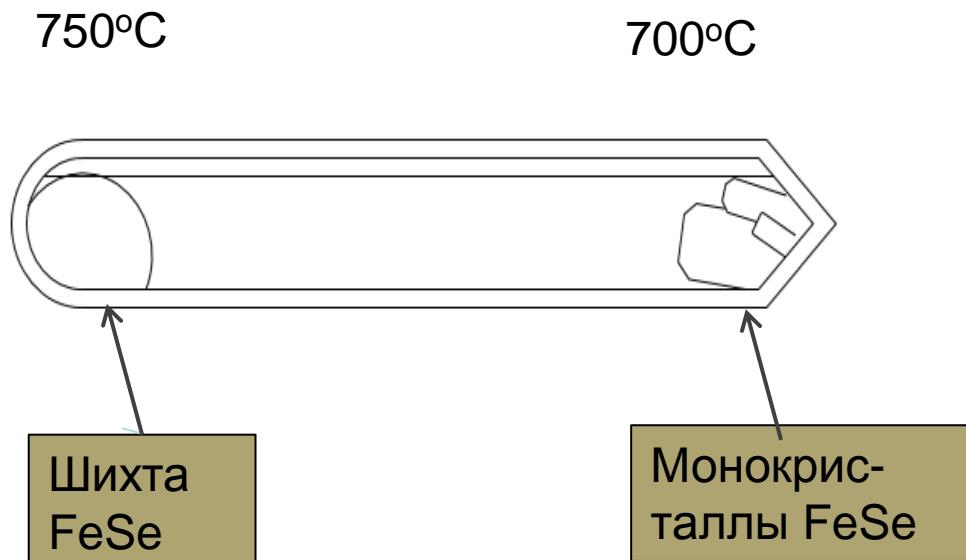
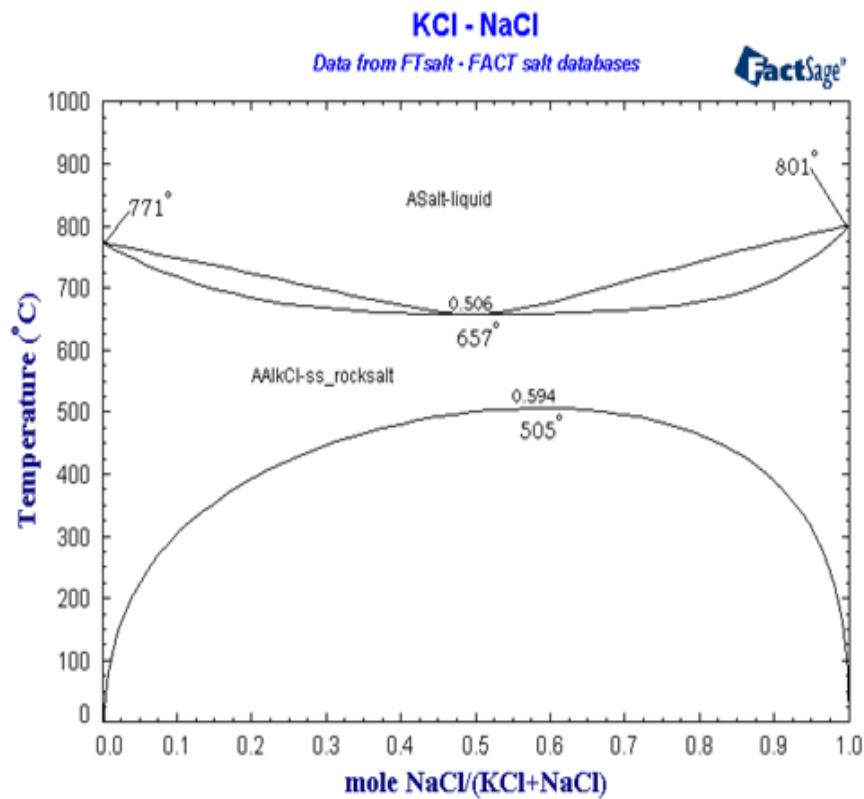
$$\text{Rb}_{0.8}\text{Fe}_2\text{Se}_2 = \text{Rb}_{0.73(6)}\text{Fe}_{1.72(3)}\text{Se}_2$$

$$\text{Rb}_{0.5}\text{Fe}_2\text{Se}_2 = \text{Rb}_{0.75(1)}\text{Fe}_{1.66(3)}\text{Se}_2 + \text{Fe}_{1+\delta}\text{Se}$$



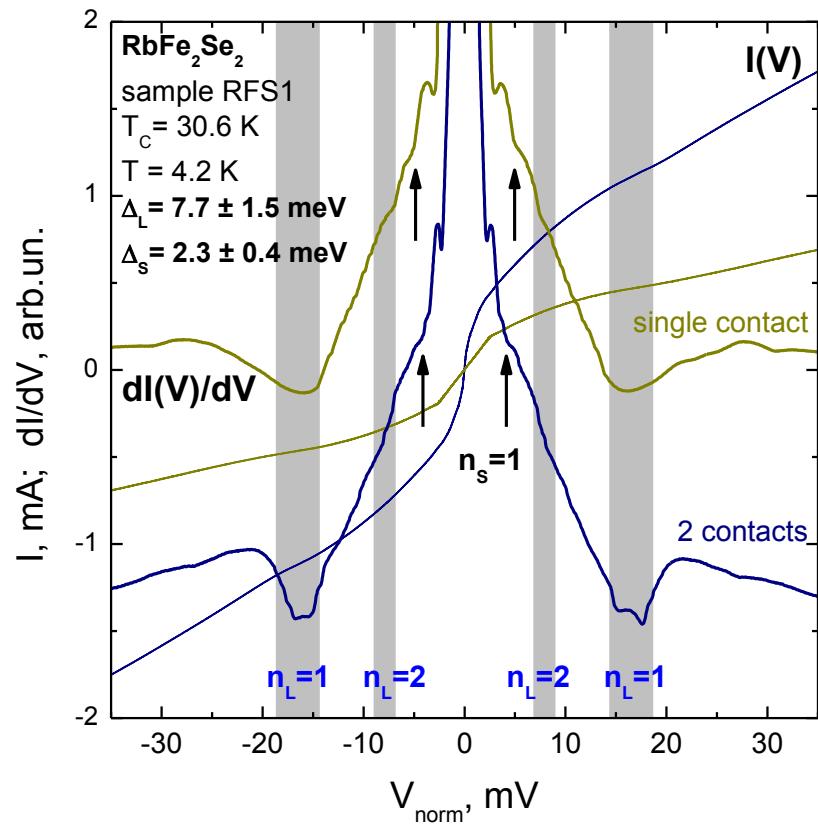
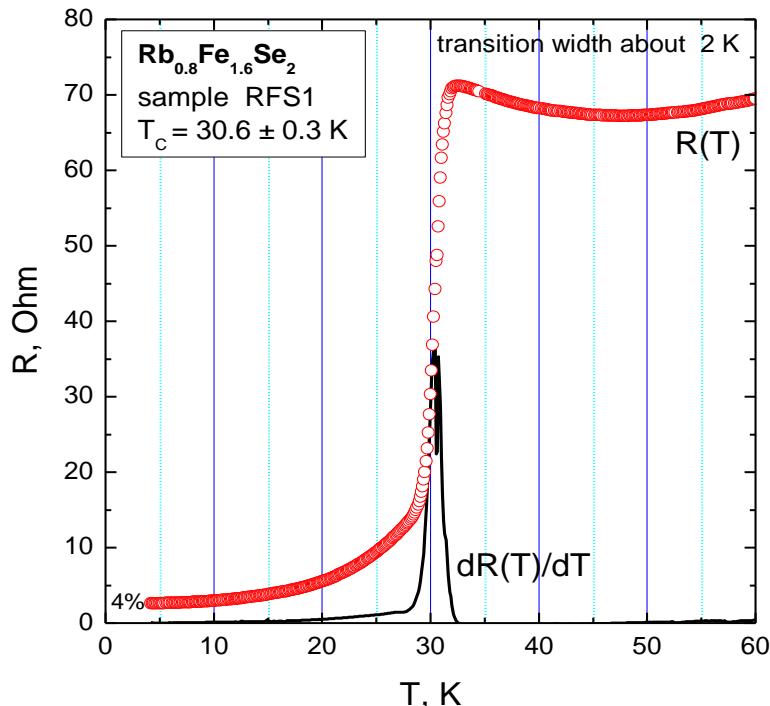
VI. Синтез образцов $(\text{Na}_z\text{K}_{1-z})_x\text{Fe}_{1-y}\text{Se}_2$ альтернативными методами

Синтез монокристаллов FeSe в эвтектических расплавах галогенидов щелочных металлов в условии стационарного градиента температур



В качестве шихты вместо FeSe загружены образцы $A_{0.8}\text{Fe}_2\text{Se}_2$, $A = \text{K}, \text{Na}, \text{Li}$

Resistivity measurements and the multiple Andreev reflections effect spectroscopy



The large and the small superconducting gap values were determined at $T = 4.2$ K:
 $\Delta_L = 7.7 \pm 1.5$ meV, $\Delta_s = 2.3 \pm 0.4$ meV. The corresponding BCS-ratios were found to be $2\Delta_L/kT_c \gg 3.52$, and $2\Delta_s/kT_c \ll 3.52$, conclude on a strong electron-boson coupling in the bands with the large gap, and on an induced superconductivity in the bands with the small gap.